



# Development and testing of refractory additively manufactured materials for fusion applications

Mykola Ialovega<sup>1</sup>,

Danah Velez<sup>1</sup>, Tyler Dabney<sup>1</sup>, Evan Willing<sup>1</sup>,

Hwasung Yeom<sup>1</sup>, Xavier Navarro-Gonzalez<sup>1</sup>,

Arkadi Kreter<sup>2</sup>,



Regis Bisson<sup>3</sup>, Thierry Angot<sup>3</sup>,



Cary Forest<sup>1</sup>, Jay Andreson<sup>1</sup>,

Kumar Sridharan<sup>1</sup> and Oliver Schmitz<sup>1</sup>

<sup>1</sup> University of Wisconsin-Madison, USA

<sup>2</sup> Forschungszentrum Jülich GmbH, Germany

<sup>3</sup> Aix Marseille Univ., PIIM, France

DCD meeting W7-X Greifswald 05/03/2024



# Outline

---

- Wisconsin HTS Axisymmetric Mirror WHAM
- Functional tantalum Plasma-Facing Components:
  - cold spray deposition
  - plasma irradiation experiments
  - deuterium retention
- Additively manufactured tungsten
- WHAM status update
- Conclusions and perspectives



# WHAM Team: We're growing

Contact: UW-Madison  
[jkanders@wisc.edu](mailto:jkanders@wisc.edu)  
or Realta Fusion  
[kieran@realtafusion.com](mailto:kieran@realtafusion.com)

Cary Forest<sup>1,12</sup>, Jay Anderson<sup>1,12</sup>, Jim Anderson<sup>4</sup>, Oscar Anderson<sup>1</sup>, Ted Biewer<sup>7</sup>, Tim Bohn<sup>1</sup>, Mike Brown<sup>11</sup>, Mirela Cengher<sup>4</sup>, Mike Clark<sup>1</sup>, Jan Egedal<sup>1</sup>, Doug Endrizzi<sup>1</sup>, Gennady Fiksel<sup>10</sup>, Kieran Furlong<sup>12</sup>, Manaure Francisquez<sup>8</sup>, Ammar Hakim<sup>8</sup>, Bob Harvey<sup>5</sup>, Mykola Ialovega<sup>1</sup>, Mi Joung<sup>13</sup>, Jeremiah Kirch<sup>1</sup>, Grant Kristofek<sup>2</sup>, Xavier Navarro-Gonzalez<sup>1</sup>, Ben Lindley<sup>1,12</sup>, John Lohr<sup>4</sup>, Ari Le<sup>6</sup>, Vladimir Mirnov<sup>1</sup>, Bob Mumgaard<sup>2</sup>, Jessica Hoffman<sup>1</sup>, Ethan Peterson<sup>3</sup>, Yuri Petrov<sup>5</sup>, Everett Penne<sup>1</sup>, Jon Pizzo<sup>1</sup>, Steve Oliva<sup>1</sup>, Charlie Moeller<sup>4</sup>, Tony Qian<sup>1,8</sup>, Kunal Sanwalka<sup>1</sup>, Oliver Schmitz<sup>1,12</sup>, Mary Severson<sup>1</sup>, Bhuvana Srinivasan<sup>9</sup>,  
**Your name here<sup>1/12</sup>**, Benjamin Terranova<sup>1</sup>, John Wallace<sup>1</sup>, Blake Wetherton<sup>6</sup>, Dennis Whyte<sup>3</sup>, Mason Yu<sup>1</sup>

<sup>1</sup> University of Wisconsin-Madison

<sup>2</sup> Commonwealth Fusion Systems

<sup>3</sup> MIT

<sup>4</sup> General Atomics

<sup>5</sup> CompxCo

<sup>6</sup> LANL

<sup>7</sup> ORNL

<sup>8</sup> PPPL

<sup>9</sup> University of Washington

<sup>10</sup> University of Michigan

<sup>11</sup> Swarthmore College

<sup>12</sup> Realta Fusion

<sup>13</sup> KSTAR NFRI



WARF  
Wisconsin Alumni Research Foundation

**REALTA**  
**FUSION**

Los Alamos  
NATIONAL LABORATORY  
tae  
FUSION POWER

**VTF**

**PPPL**



**NFRI**  
국가핵융합연구소

**M**  
UNIVERSITY OF  
MICHIGAN

**arpa-e**



**Commonwealth  
Fusion Systems**

**HIT PSFC**

**COMPX**

**OAK RIDGE**  
National Laboratory



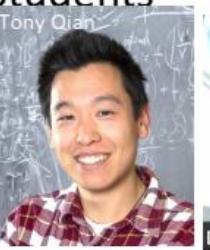
Wisconsin HTS Axisymmetric Mirror

# UW Physics, Engineering Physics and Physical Sciences Lab

## Students



Tony Qian



## Post Docs



Douglass Endrizzi



Jan Egedal



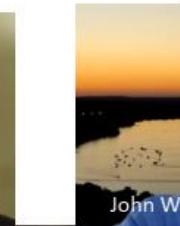
Oliver Schmitz



Cary Forest



Jay Anderson



John Wallace



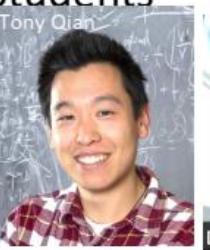
Steve Oliva



Jeremiah Kirch



Kunal Sanwala



Tony Qian



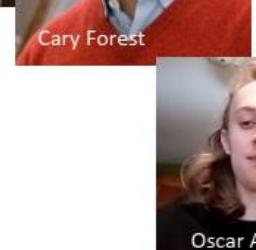
Douglass Endrizzi



Mike Brown



Kieran Furlong,  
CEO Realta Fusion



Oscar Anderson



Everett Penne



Mike Clark



Jess Hoffman



REST IN PEACE

## MIT and CFS



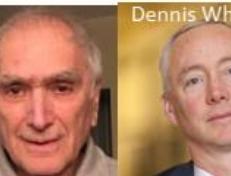
Sergey Kuznetsov



Alexey Radovinsky



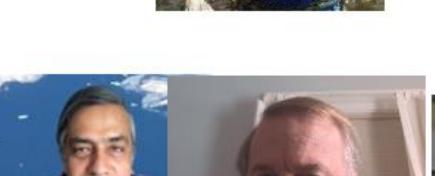
David Melchior



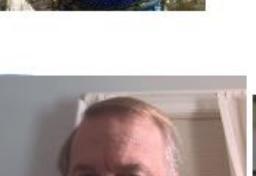
Dennis Whyte



Alex Zhukovsky



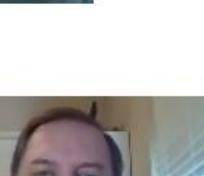
Abhay Ram



John Wright



Dylan  
Copeland



Yuri Petrov

## GA ECH



Charles Moller



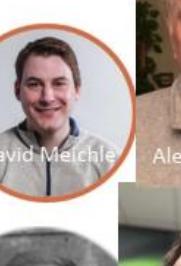
John Lohr



Grant Kristofek



Blake Mitchell



Dan Nash



Nick Kelton



Bob Mumgaard



Mark Stowell



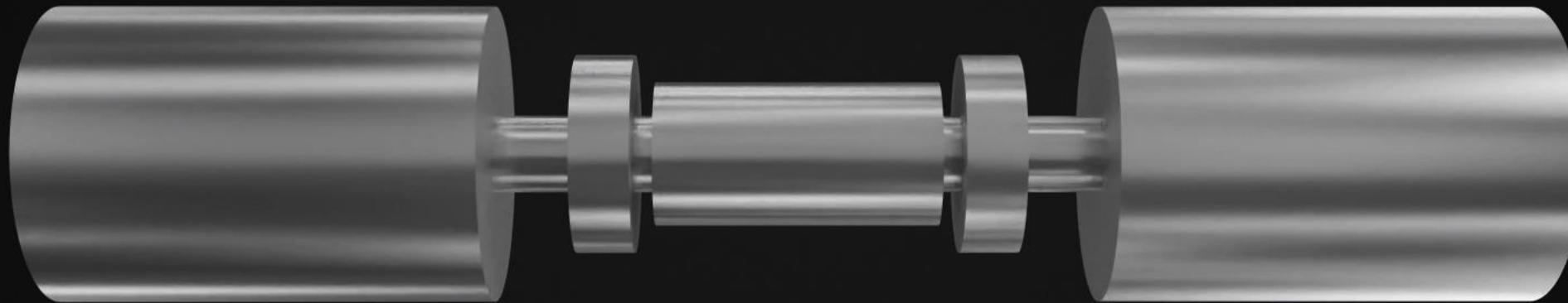
Atul Kumar



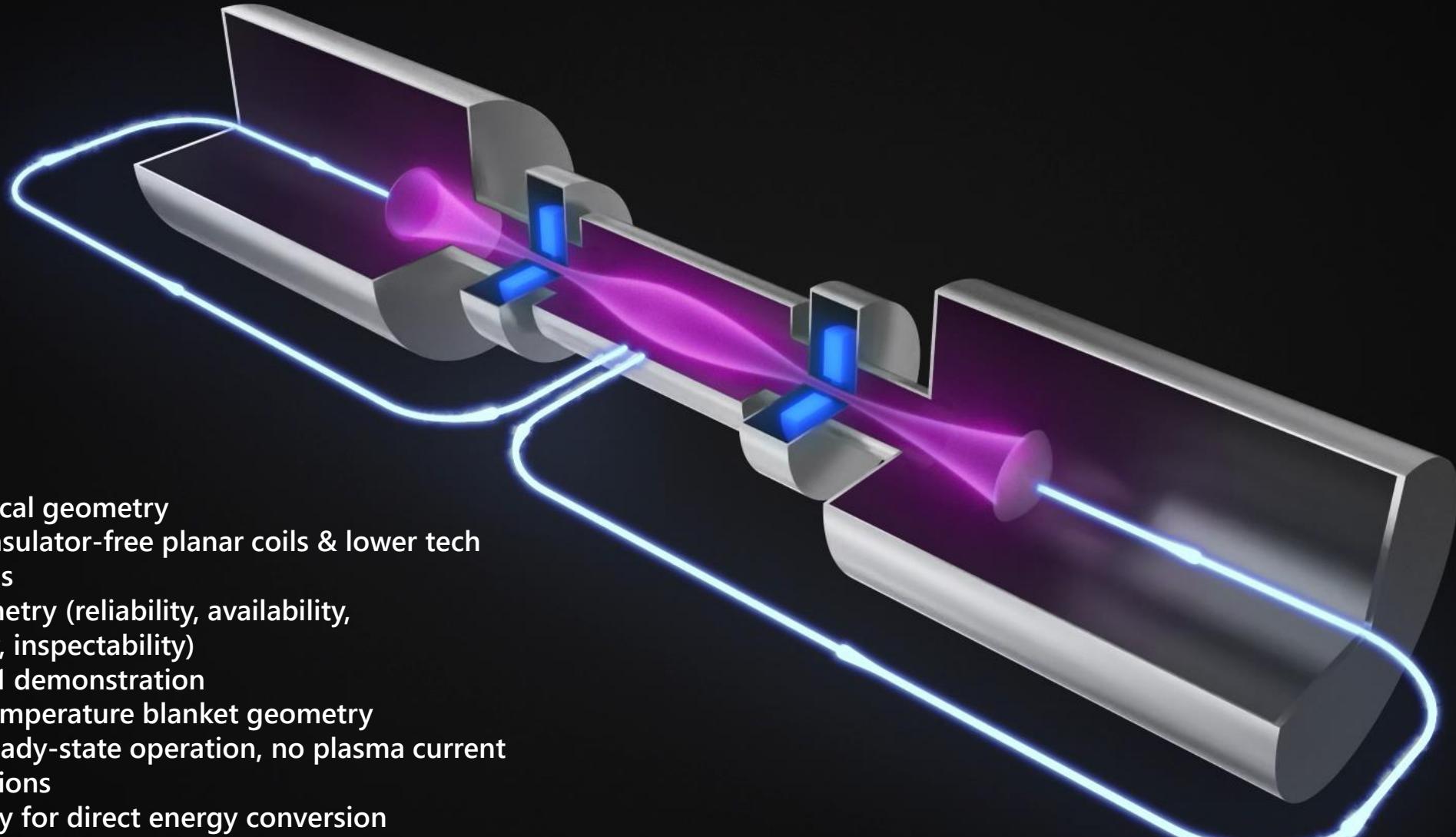
Juan  
Capoerla Marquez



# Magnetic mirrors concept



# Magnetic mirrors concept



- Simple cylindrical geometry
  - high-field insulator-free planar coils & lower tech central cell coils
  - Linear geometry (reliability, availability, maintainability, inspectability)
- Less costly Q>1 demonstration
- Simple high-temperature blanket geometry
- Intrinsically steady-state operation, no plasma current and no disruptions
- Open geometry for direct energy conversion
- Allows low-tritium use materials testing, component testing fuel cycle demonstration etc.

# WHAM goals

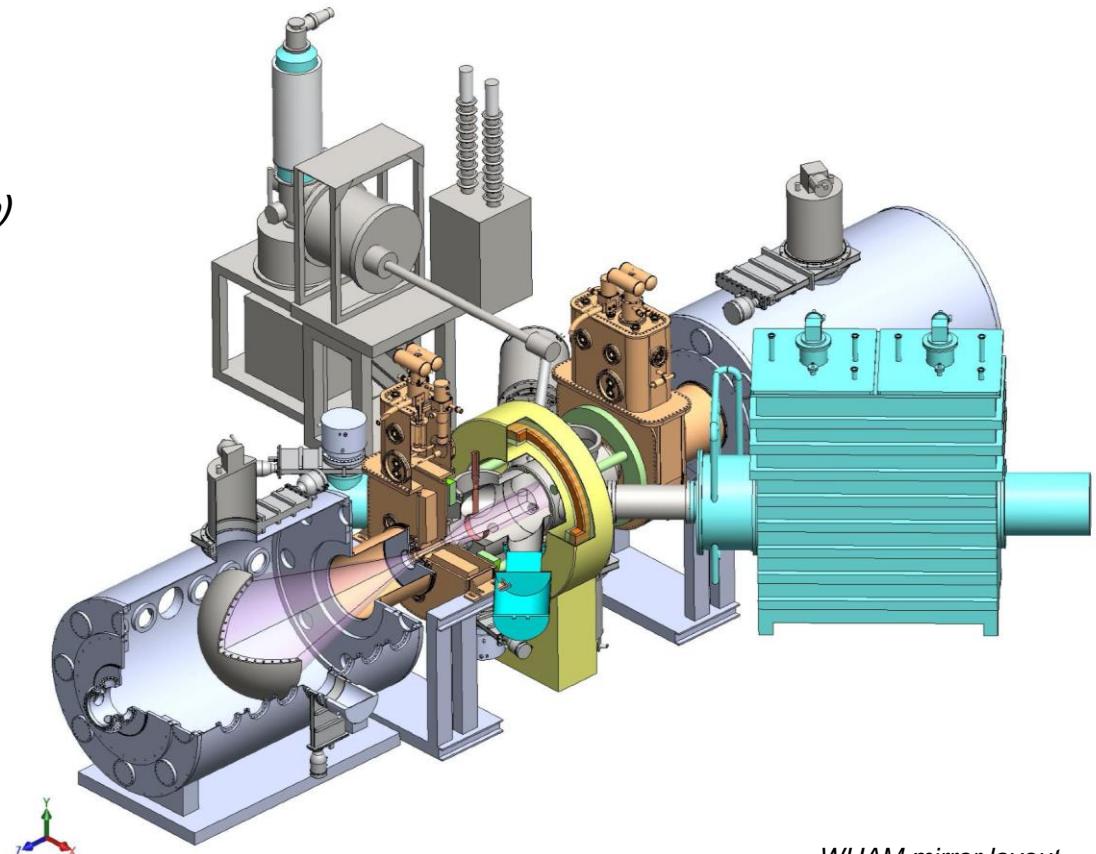
WHAM (Wisconsin HTS Axisymmetric Mirror) experiment to demonstrate  
“Breakthrough” physics for fusion energy on tandem mirror path.

Physics Missions:

1. *Confine MHD stable, high Te plasma in axisymmetric mirror (endcell prototype)*
  - Requires simultaneous MHD stability + high power ECH
  - Requires negligible cross-field transport, axial loss only
2. *Demonstrate novel in-situ ion acceleration*
  - Requires high power RF system and NBI
  - Verify wave damping characteristics (on which particles, where in plasma column)
  - Verify sufficient *kinetic stability* to reap confinement benefits of high energy ions

Technology Missions: (intertwined with physics goals)

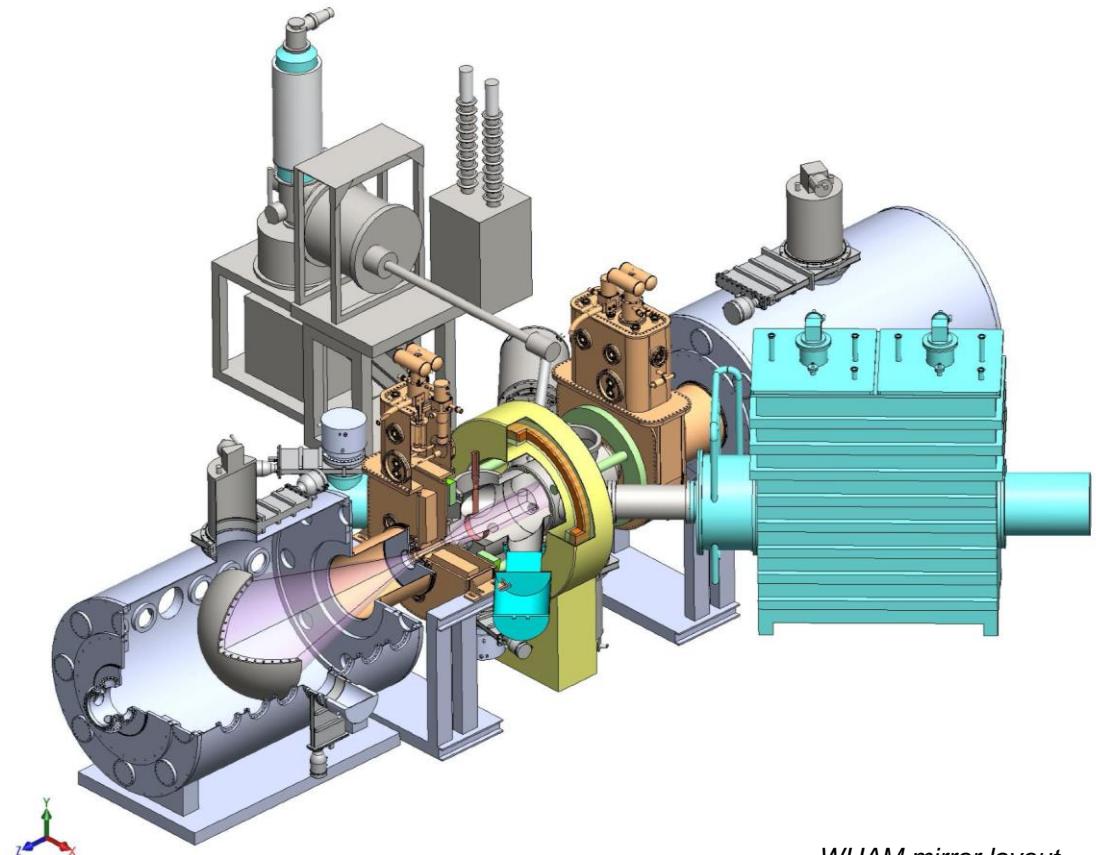
1. *Actualize HTS mirror reactor coils: CFS.*
  - Build 17 T, 5.5 cm bore HTS coils; design 25 T, 32 cm.
2. *Demonstrate expander particle handling techniques*
  - Create low neutral pressure, low recycling, low sputtering (in short pulse length) with high ion energies.



WHAM mirror layout

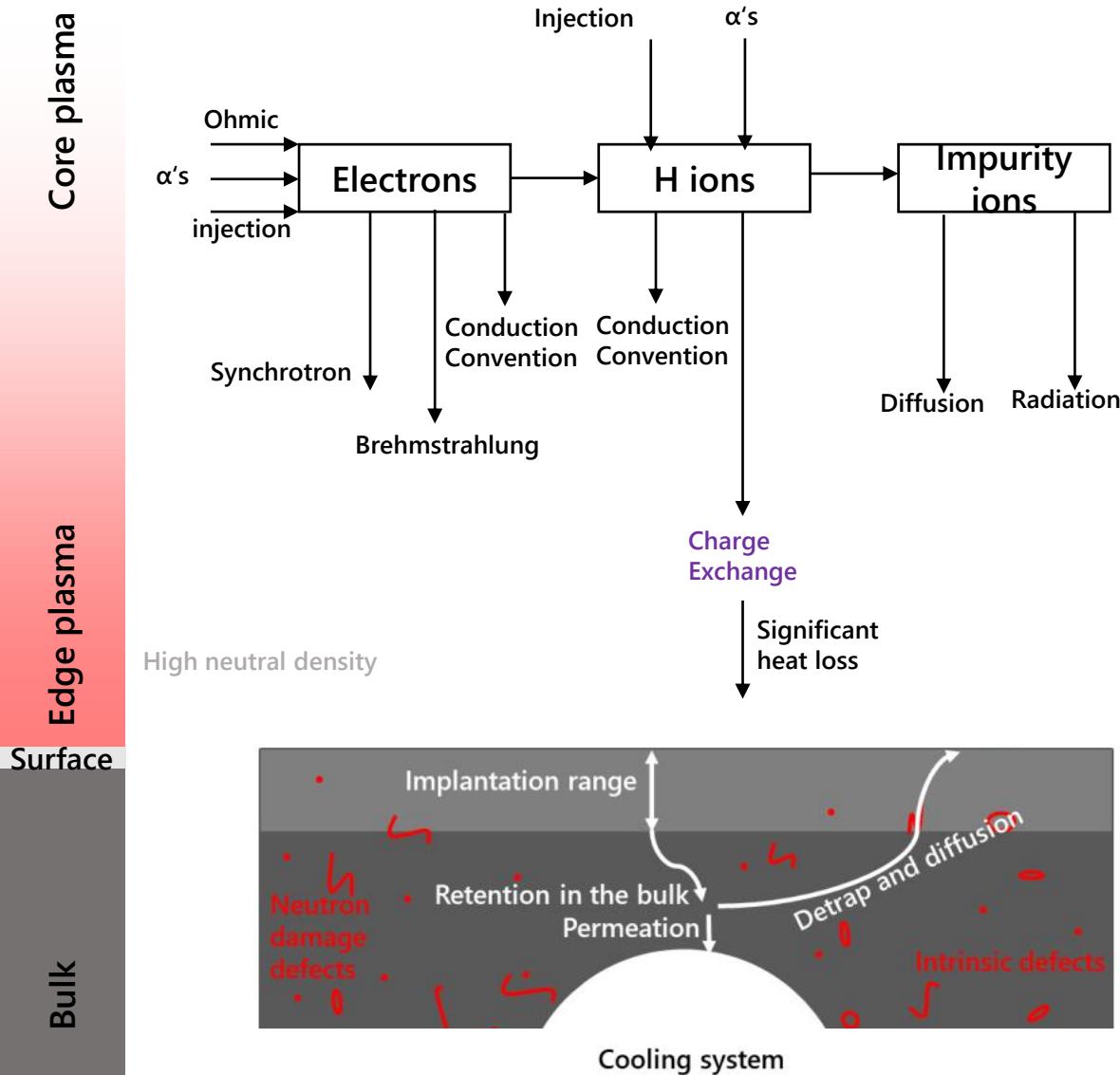
# WHAM parameters

	WHAM
$Q_{DT}$	$\sim 0.05$
$\beta$	0.3
$L_p$ (m)	2.0
$a_0$ (m)	0.12
$a_M$ (m)	0.025
$R_M$	20
$B_0$ (Tesla)	0.85
$B_M$ (Tesla)	17
$E_{NBI}$ (keV)	30
$P_{NBI}$ (MW)	< 1
$T_i$ (keV)	17
$T_e$ (keV)	2.5
$n_{20}$ ( $m^{-3}$ )	0.3
$\tau_p$ (sec) <sup>†</sup>	0.15
$nT\tau$ ( $10^{20}m^{-3}\cdot keV\cdot sec$ )	0.7
pulse length (sec)	< 2
kA·m of HTS tape	$\sim 10^4$
Cost of tape (\$M)	$\sim 1$



WHAM mirror layout

# A key to success: minimizing charge-exchange events



Charge exchange (CX) – exchange of an electron between a positive ion and a neutral particle. The ion in the plasma edge is converted to a neutral which is not confined.

Effective control of hydrogen isotopes (HIs) partial pressure in the plasma edge in small fusion devices is important to limit energy losses associated with the charge exchange events.

CX losses is the key issue for small devices like WHAM

=>

Functional first wall interface for HIs neutral absorption

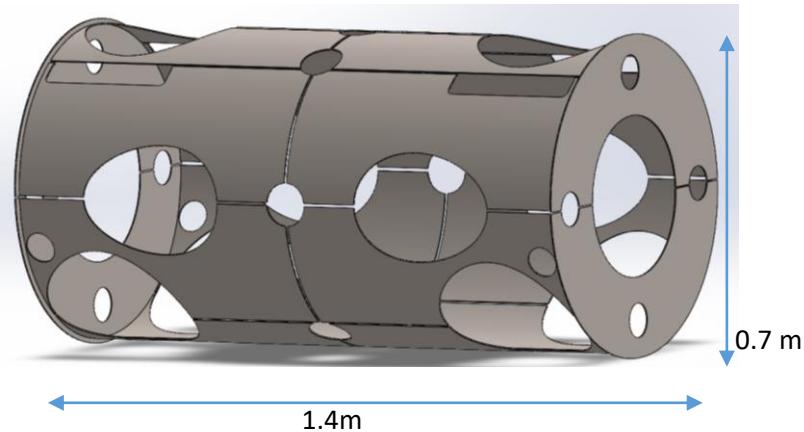
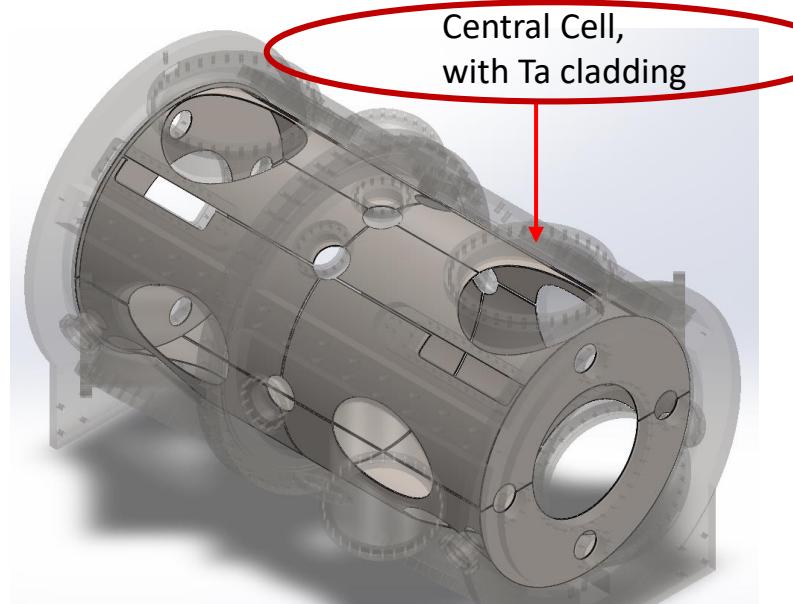
# Functional first-wall concept

- Functional first wall for effective neutrals and energetic H (D) ions absorption at certain temperature and pressure conditions.

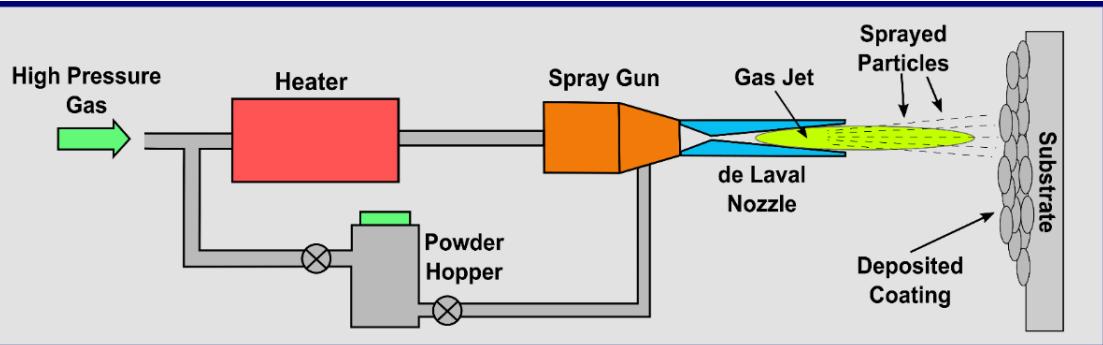
73	180.948
<b>Ta</b>	
Tantalum	
[Xe] 4f <sup>14</sup> 5d <sup>3</sup> 6s <sup>2</sup>	
Transition Metals	

- excellent hydrogen storage capacity,
- high threshold for particle sputtering,
- excellent room-temperature ductility,
- unique corrosion properties

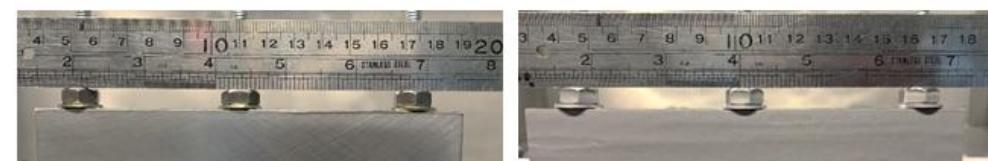
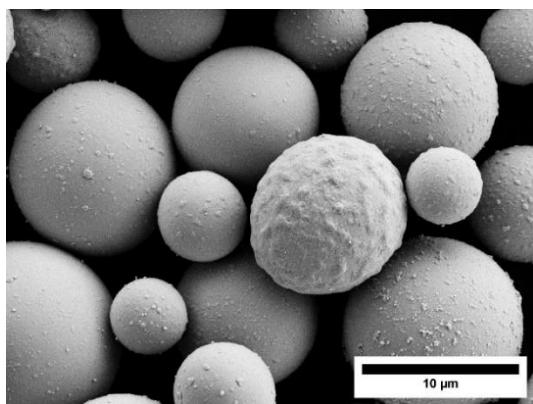
	Ti	Mo	Ta	W
Melting point, K	1941	2896	3123	3687
Thermal conductivity (293 K), W/(m K)	22.4	138	57.4	182
Sputtering TRH (by D), eV	33.2	90	315	244
Hydrogen retention	high	lower	high	low



# Cold-spray deposition process



- On-site component manufacturing & repair
- Potentially in-situ
- Deposition of any thickness
- Various shapes (for stellarator divertor, specially shaped vacuum chambers)
- Commercially available



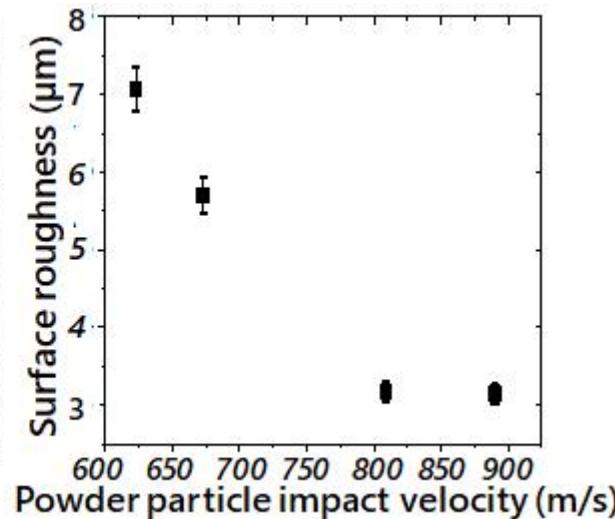
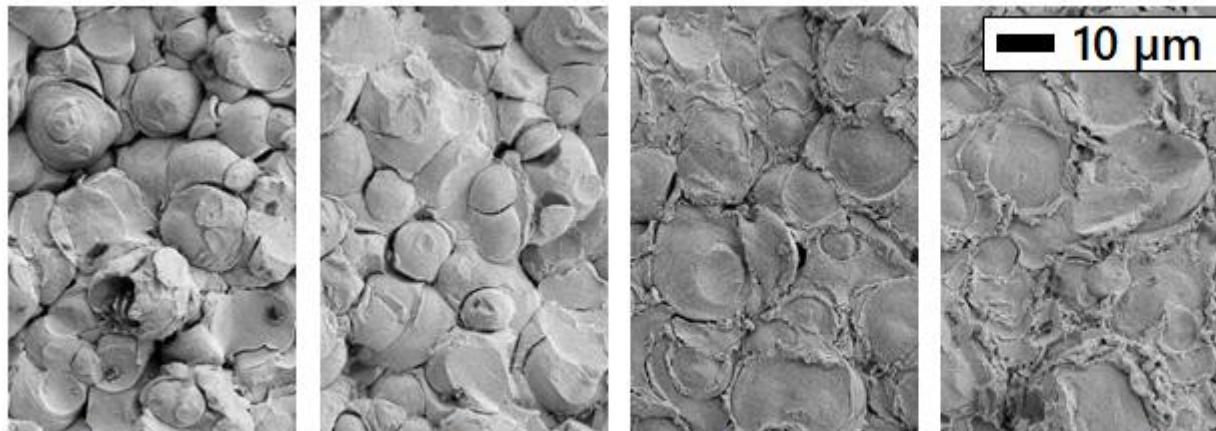
SS substrate before deposition

Ta(CS) N<sub>2</sub>, 1023 K, 4.0 MPa,  
400 mm/s;  $V_p = 673 \text{ m/s}$ ,  $T_p = 590 \text{ K}$

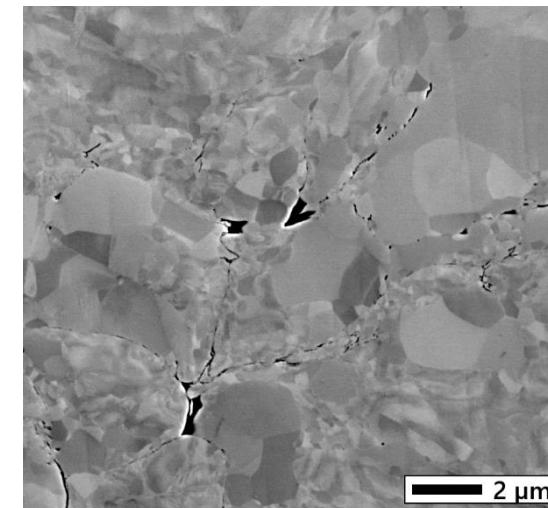
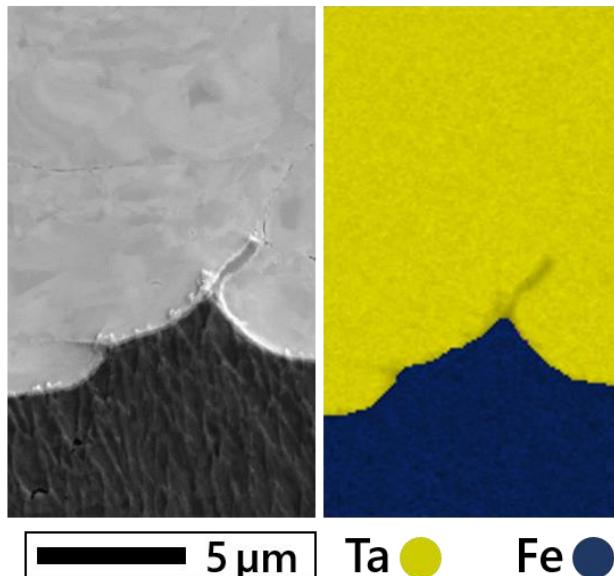
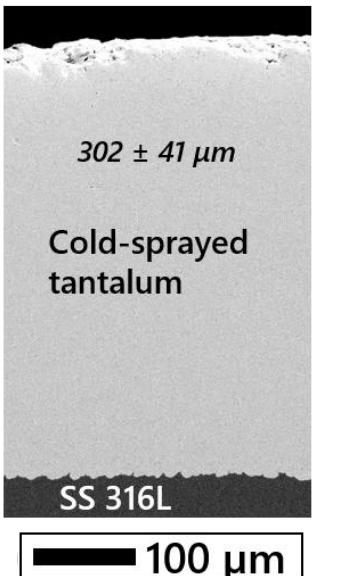


# Characterization of Ta cold spray coatings

Surface



Cross-section

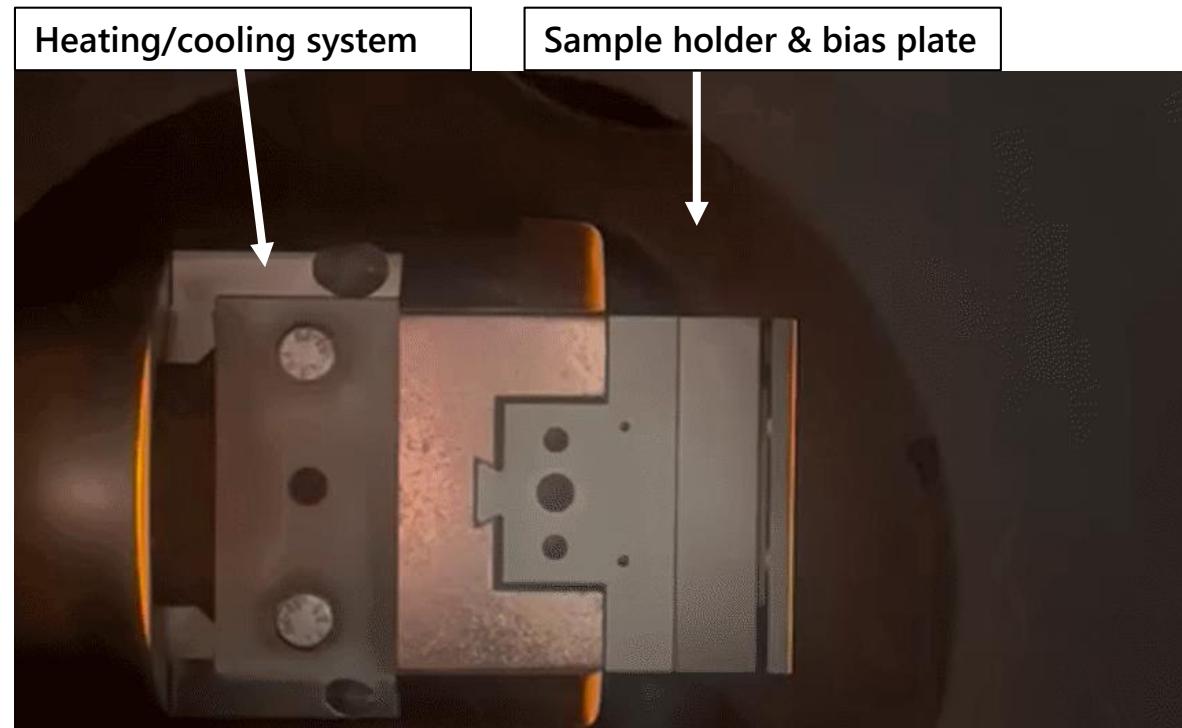


- Particle velocity determines particle bonding, surface and bulk morphology, stress, etc
- Dense, well-adhered coatings
- Controlled porosity level

# D irradiation tests on cold spray Ta coatings

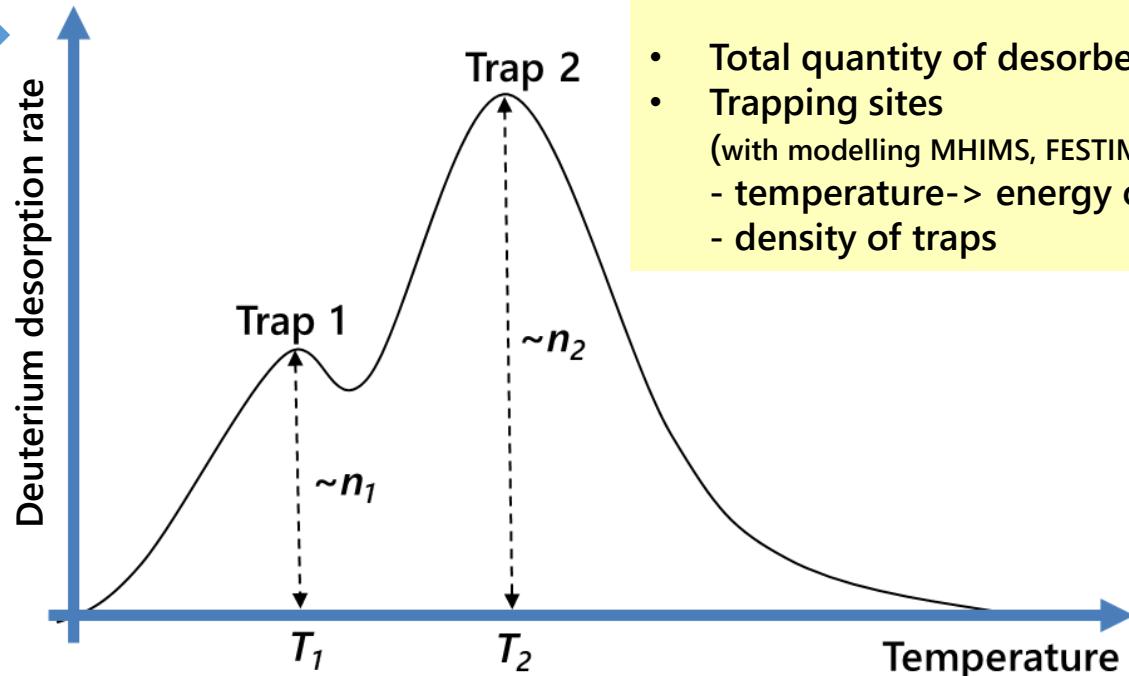
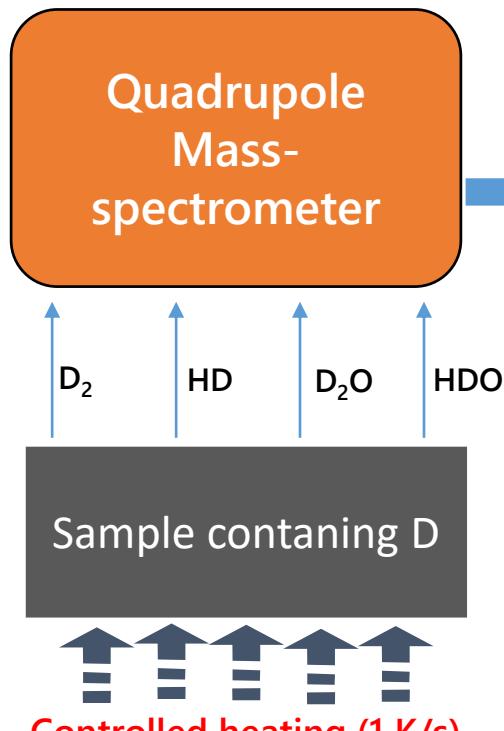
Choice of plasma-exposure parameters:

- D plasma
- $E_{kin}=95$  eV: no erosion, no damage due to the exposure
- Constant (high) flux  $1.6e21$  D m $^{-2}$ s $^{-1}$
- Incident fluence [ $1.6e22$ ,  $3e25$ ] D m $^{-2}$
- Surface temperature [420,950] K



# Experimental methods: Thermodesorption

Main technique: Thermodesorption Spectrometry (TDS)



TDS allows to determine:

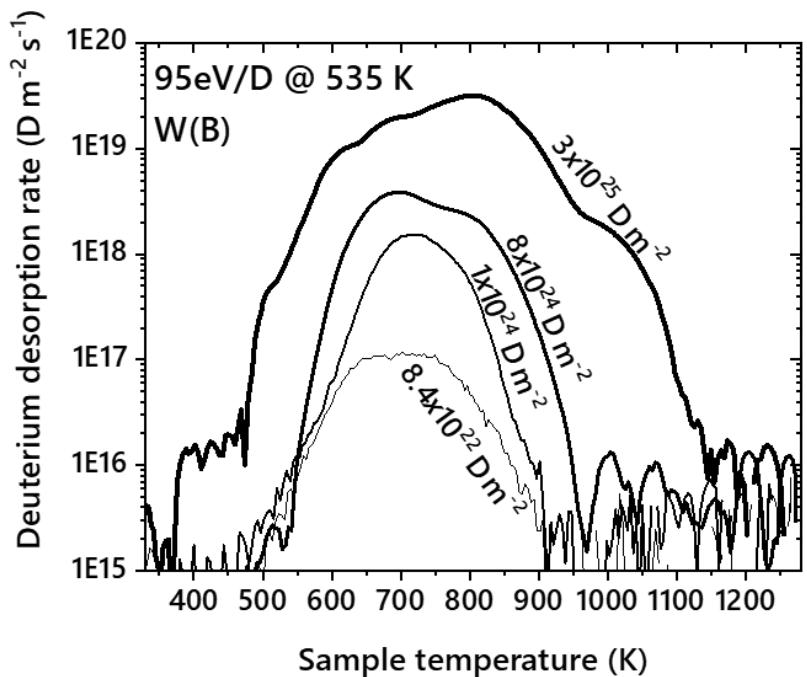
- Total quantity of desorbed species
- Trapping sites  
(with modelling MHIMS, FESTIM, etc.):
  - temperature  $\rightarrow$  energy of traps
  - density of traps

$$I(T) \sim -\frac{d\theta}{dt} = v(\theta, T) \cdot \theta^n \cdot \exp\left(\frac{-E_{des}(\theta, T)}{k_B T}\right)$$

$v$  Frequency factor  
 $\theta$  Instantaneous coverage  
 $n$  Kinetic desorption order ( $0 \leq n \leq 2$ )  
 $E_{des}$  Activation energy to desorption

Polanyi-Wigner Equation

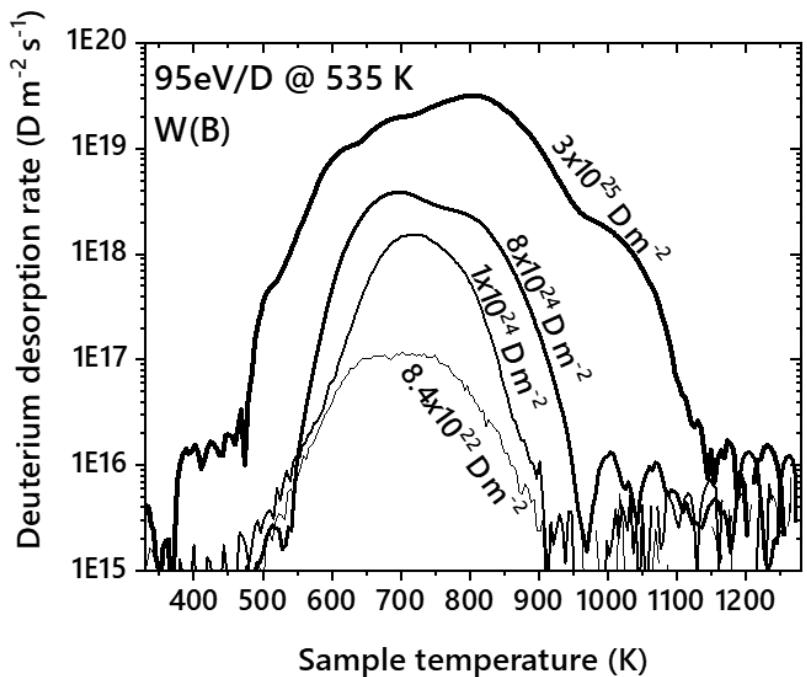
# D retention in reference tungsten (bulk)



Desorption can be expressed:

$$-\frac{d\Theta}{dT} = \frac{\nu_0}{\beta} \Theta^m \exp\left(-\frac{E_d}{k_B T}\right), \quad \Theta = \frac{c_{surface}}{n_{surface}}$$

# D retention in reference tungsten (bulk)

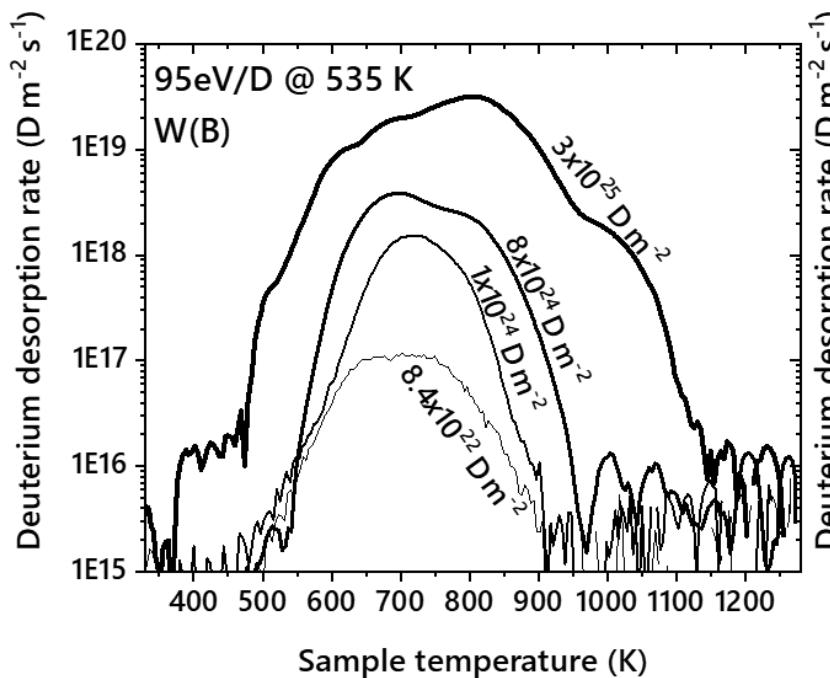


First order desorption kinetics

$$\frac{\beta}{k_B T_m^2} = \frac{\nu_0}{E_d} \exp\left(-\frac{E_d}{k_B T_m}\right) \quad E_d = 2.0 \text{ eV}$$

De-trapping from grain boundaries and voids

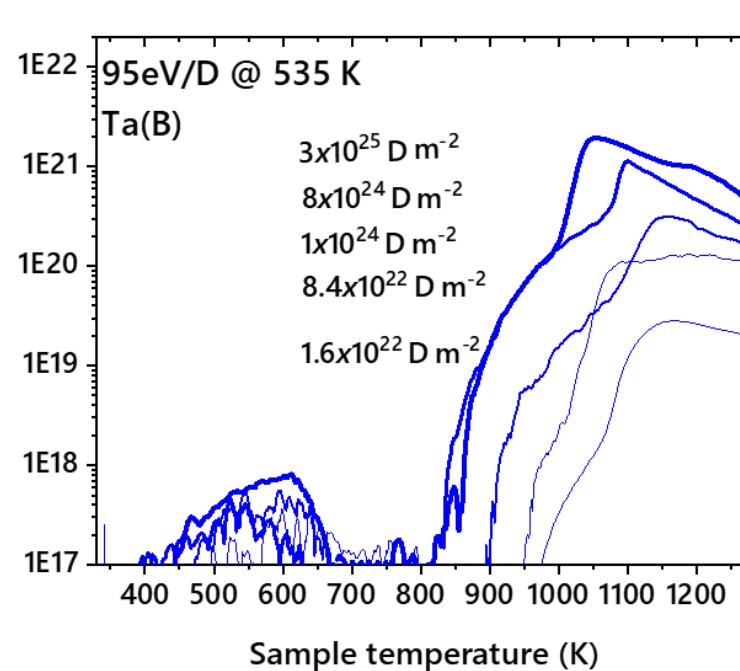
# D retention in reference tantalum (bulk)



First order desorption kinetics

$$\frac{\beta}{k_B T_m^2} = \frac{\nu_0}{E_d} \exp\left(-\frac{E_d}{k_B T_m}\right) \quad E_d = 2.0 \text{ eV}$$

De-trapping from grain boundaries and voids

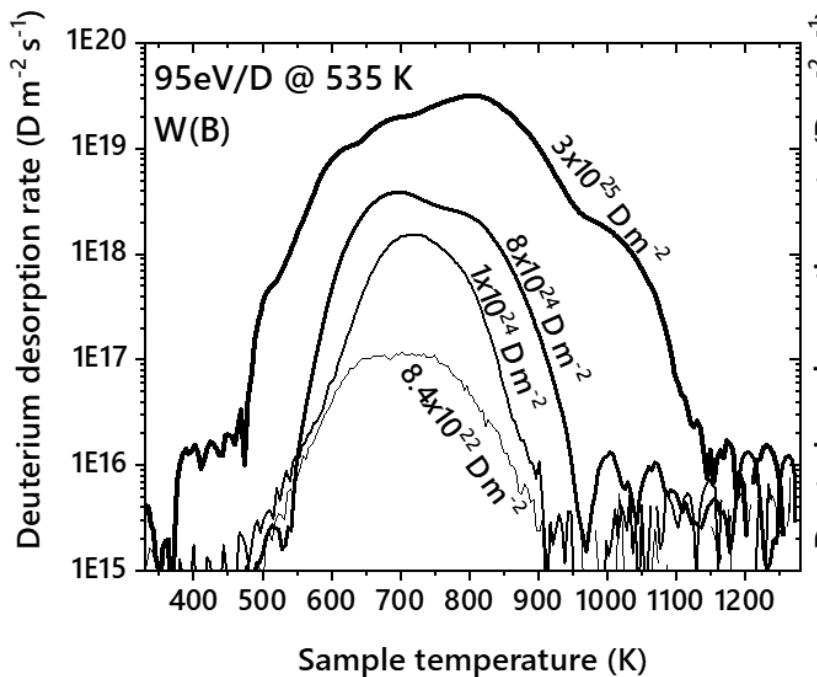


Second order desorption kinetics

$$\frac{\beta}{k_B T_m^2} = 2\Theta \frac{\nu_0}{E_d} \exp\left(-\frac{E_d}{k_B T_m}\right) \quad E_d = 3.15 \text{ eV}$$

Dedicated modelling is necessary to identify the trapping mechanisms

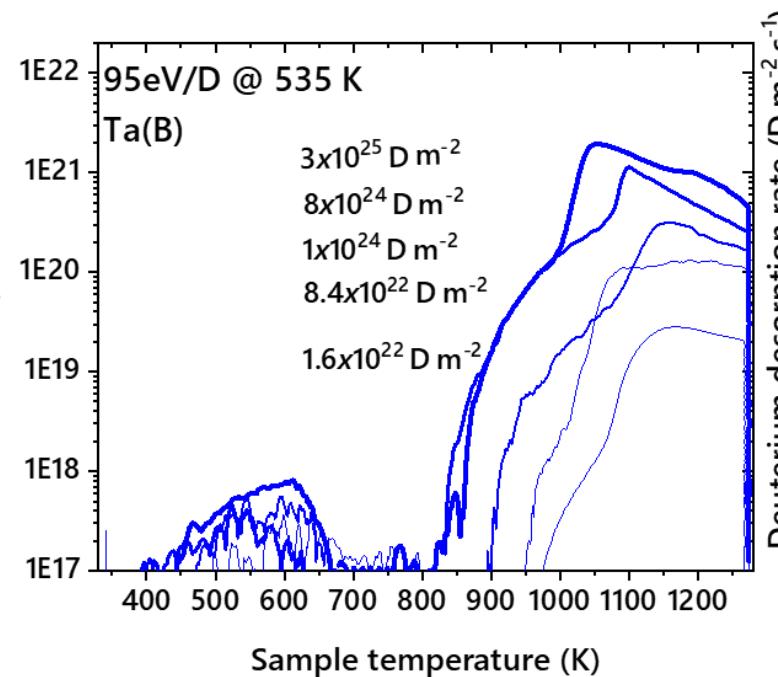
# D retention in tantalum cold spray



First order desorption kinetics

$$\frac{\beta}{k_B T_m^2} = \frac{\nu_0}{E_d} \exp\left(-\frac{E_d}{k_B T_m}\right) \quad E_d = 2.0 \text{ eV}$$

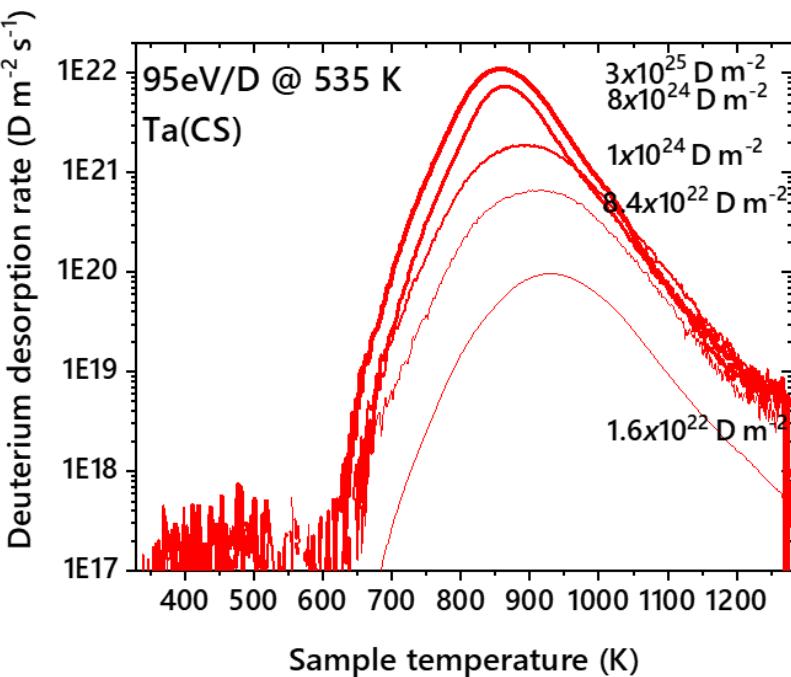
De-trapping from grain boundaries and voids



Second order desorption kinetics

$$\frac{\beta}{k_B T_m^2} = 2\Theta \frac{\nu_0}{E_d} \exp\left(-\frac{E_d}{k_B T_m}\right) \quad E_d = 3.15 \text{ eV}$$

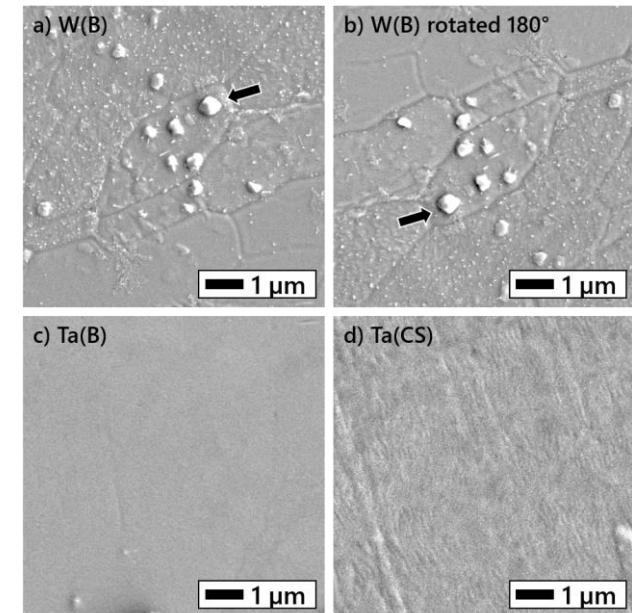
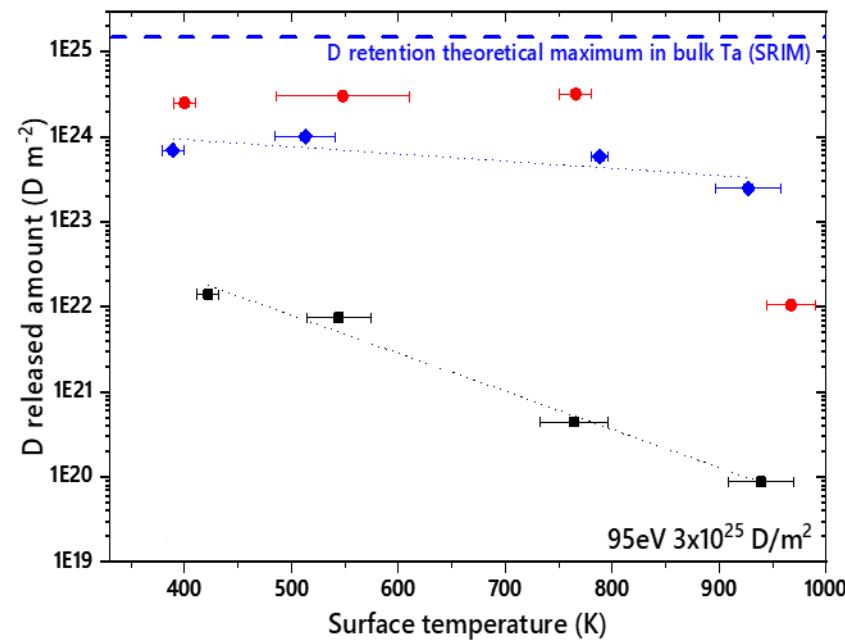
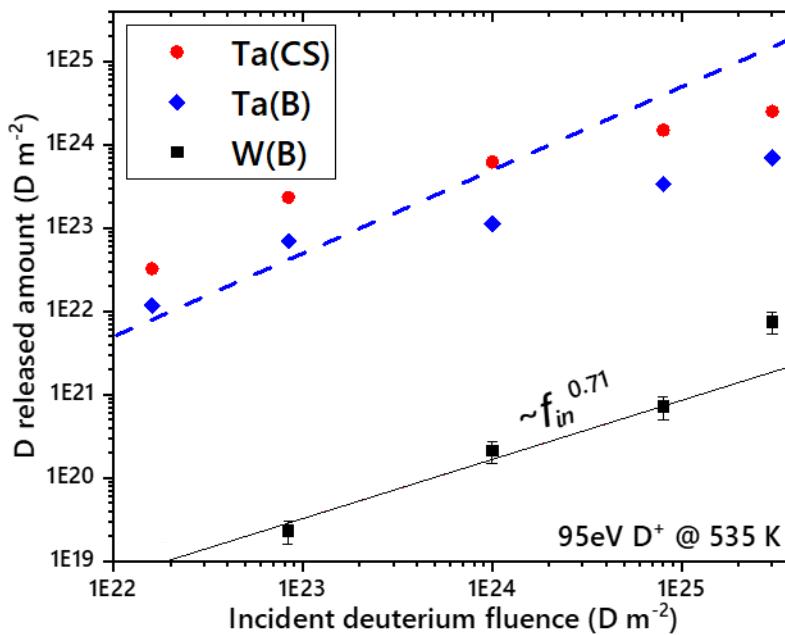
Dedicated modelling is necessary to identify the trapping mechanisms



Second order desorption kinetics

$$E_d = 2.56 \text{ eV}$$

# Enhanced D retention in tantalum cold spray

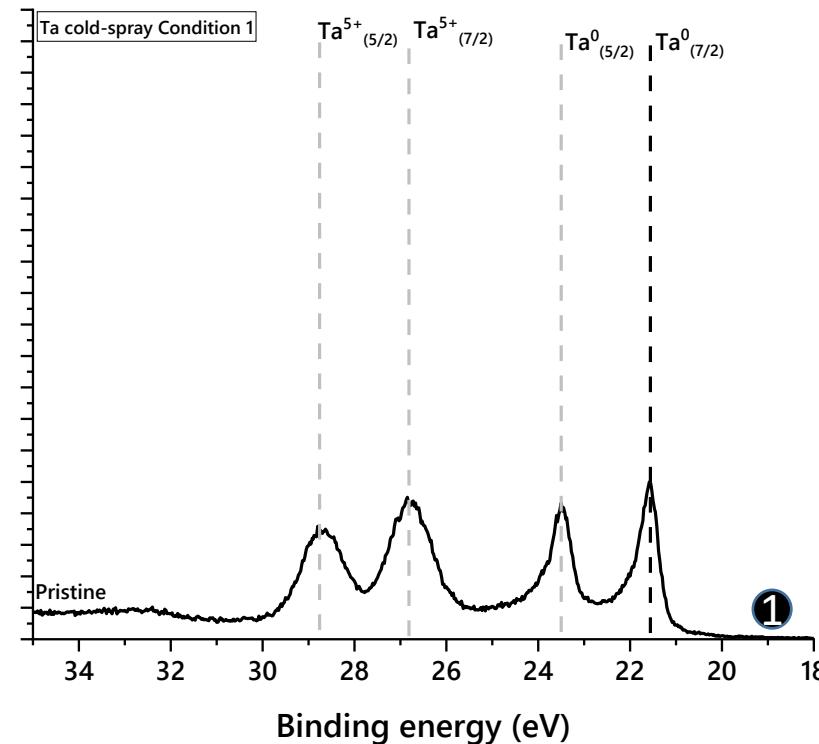


- 2 orders of magnitude greater retention in Ta than in W
- 3.5 times enhanced retention in cold spray compared to bulk Ta
- Retention is independent on the Ta substrate temperature up to  $\sim 800$  K
- Morphology of the coatings may promote D diffusion/outgassing
- Blisters on W surface may play a role in D trapping
- No surface modification on Ta samples

# Surface composition of pristine cold-spray Ta

X-ray photoelectron spectroscopy (XPS) – compositional analysis at the surface  
XPS is not able to detect D directly

Pristine cold-sprayed sample:  
① Tantalum pentoxide at the surface  
• Natural passivating layer



# D implantation: Near-surface composition evolution

X-ray photoelectron spectroscopy (XPS) – compositional analysis at the surface

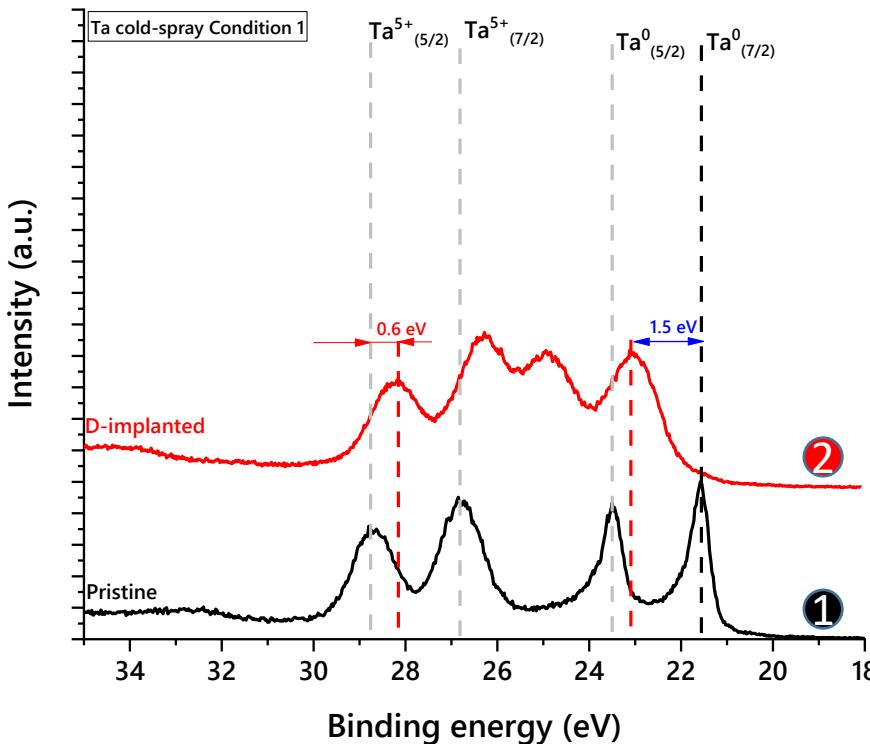
XPS is not able to detect D directly

Pristine cold-sprayed sample:

- ① Tantalum pentoxide at the surface
  - Natural passivating layer

② D-implanted tantalum:

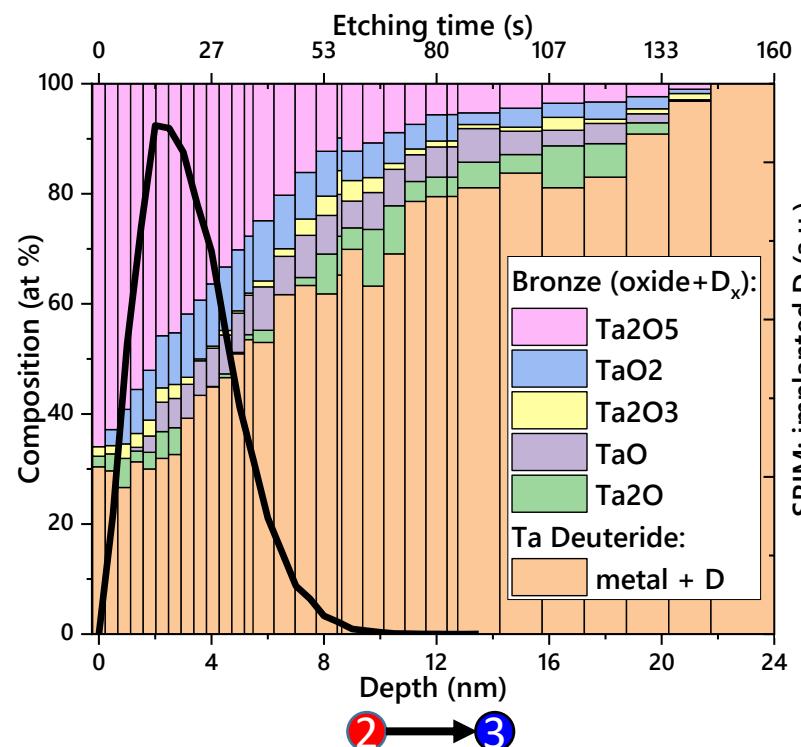
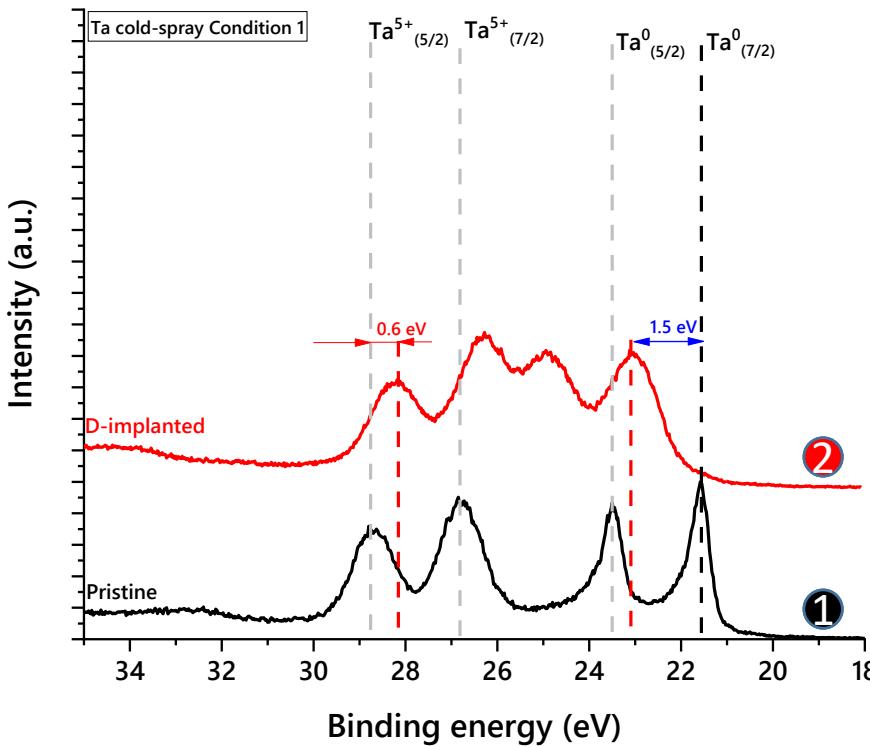
- Peak shift:  
lower oxidation states?  
possible chemical interaction between Ta, D  
and O results in the formation of  $TaD_xO_y$
- Peak shift for Ta → Tantalum Deuteride?



# D implantation depth estimation

X-ray photoelectron spectroscopy (XPS) – compositional analysis at the surface

XPS is not able to detect D directly



Pristine cold-sprayed sample:

- ① Tantalum pentoxide at the surface
  - Natural passivating layer

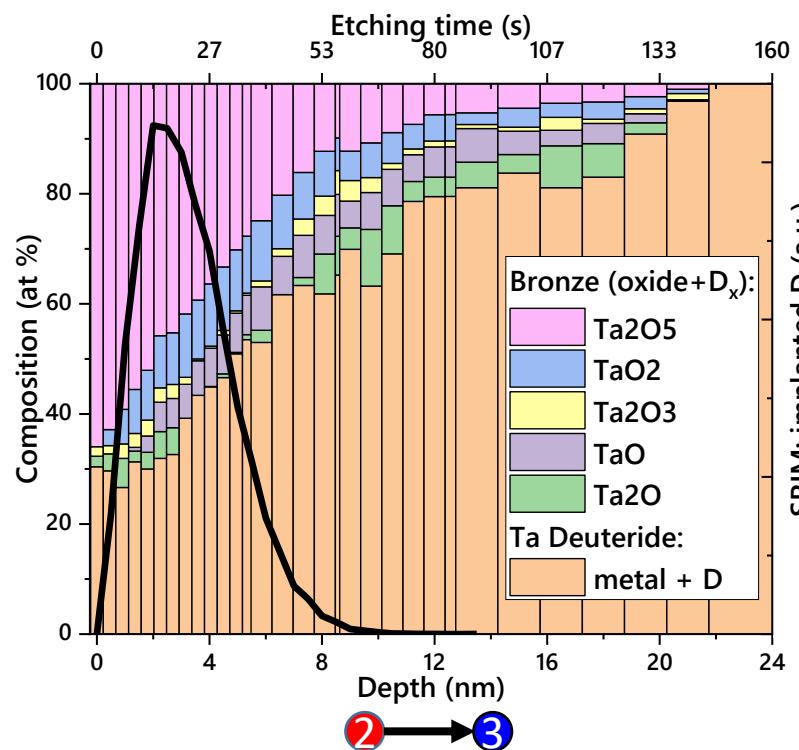
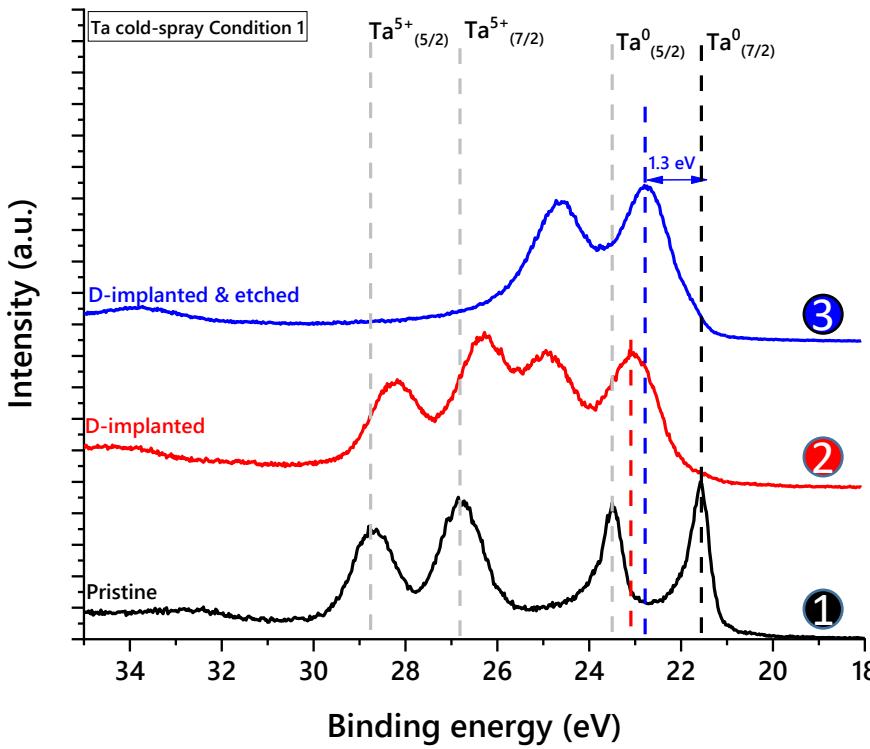
② D-implanted tantalum:

- Peak shift:  
lower oxidation states?  
possible chemical interaction between Ta, D  
and O results in the formation of  $TaD_xO_y$
- Peak shift for Ta → **Tantalum Deuteride?**

③ Etching (2 keV Ar, 0.15 nm/sec) removes the oxide

# D implantation depth estimation

X-ray photoelectron spectroscopy (XPS) – compositional analysis at the surface  
 XPS is not able to detect D directly



Pristine cold-sprayed sample:

- ① Tantalum pentoxide at the surface
  - Natural passivating layer

② D-implanted tantalum:

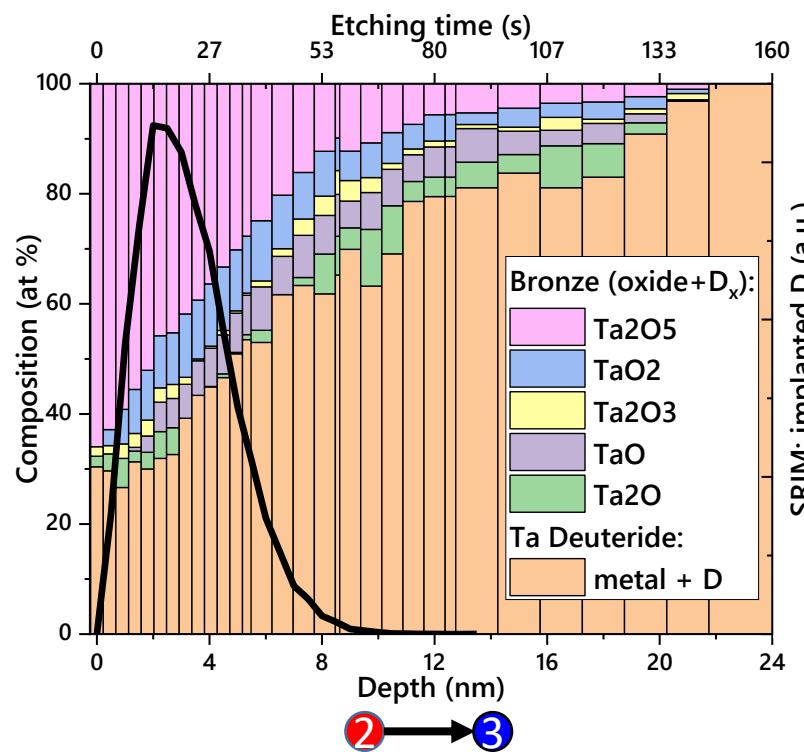
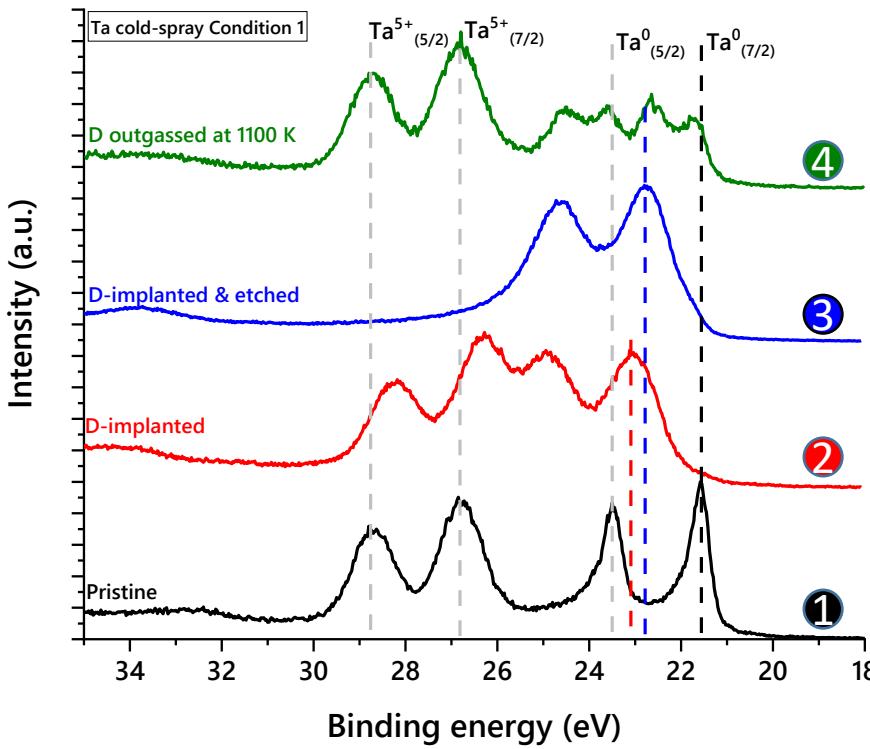
- Peak shift:  
lower oxidation states?  
possible chemical interaction between Ta, D and O results in the formation of  $\text{TaD}_x\text{O}_y$
- Peak shift for Ta → Tantalum Deuteride?

③ Etching (2 keV Ar, 0.15 nm/sec) removes the oxide

- D implantation depth estimate: < 24 nm?

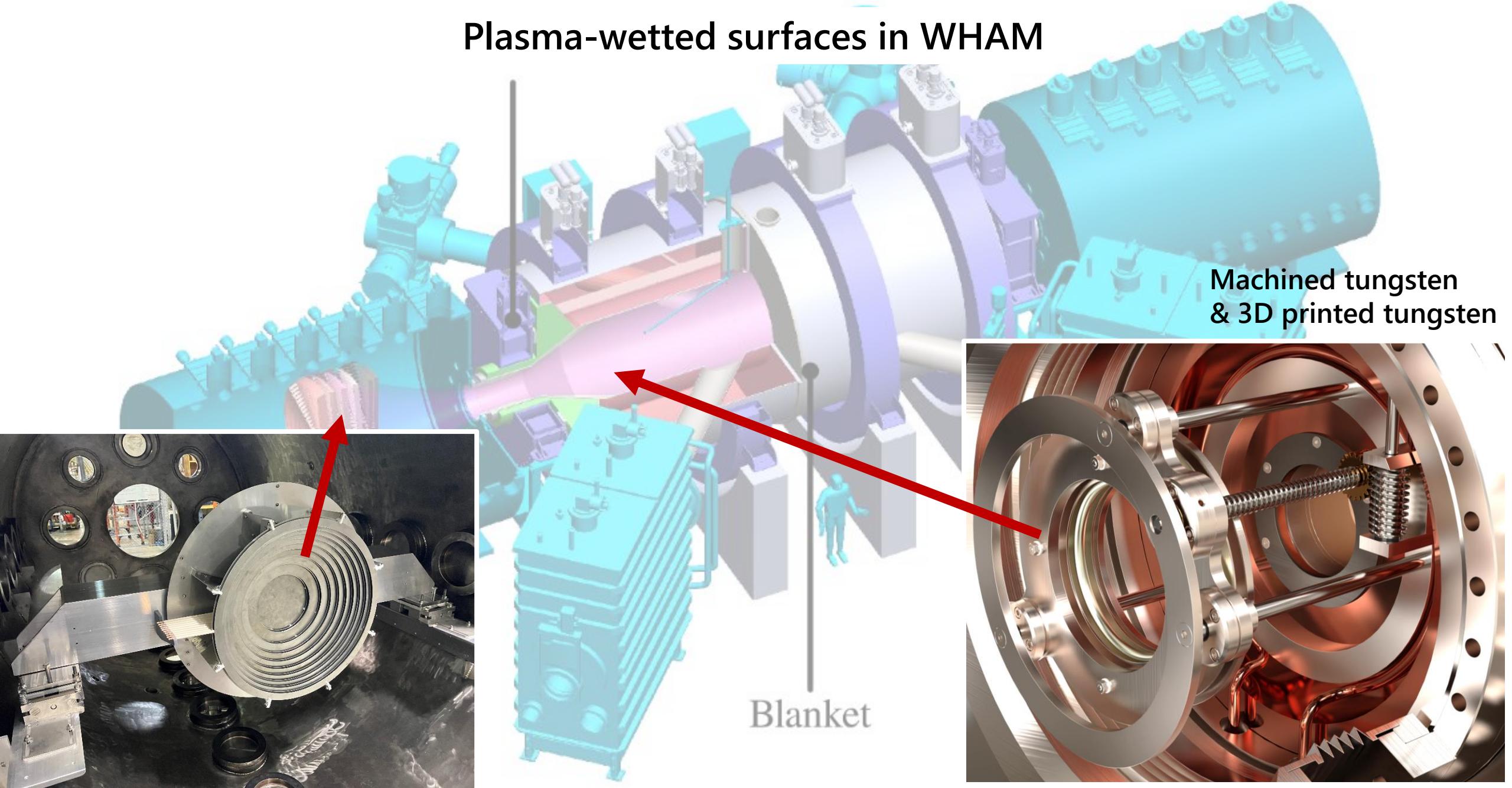
# D outgassing: recovery of metallic Ta

X-ray photoelectron spectroscopy (XPS) – compositional analysis at the surface  
XPS is not able to detect D directly



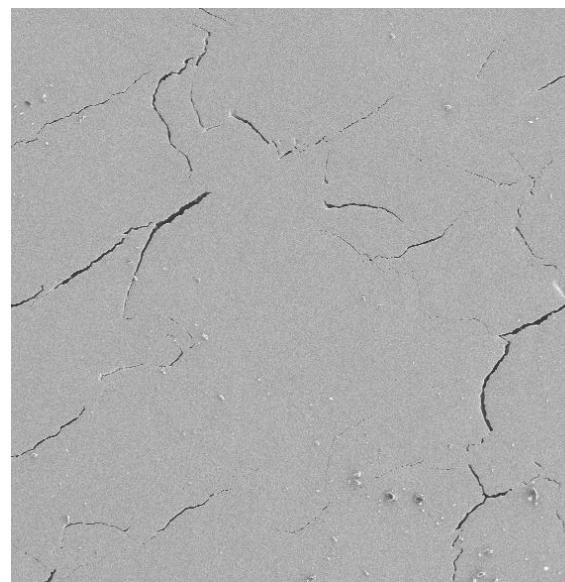
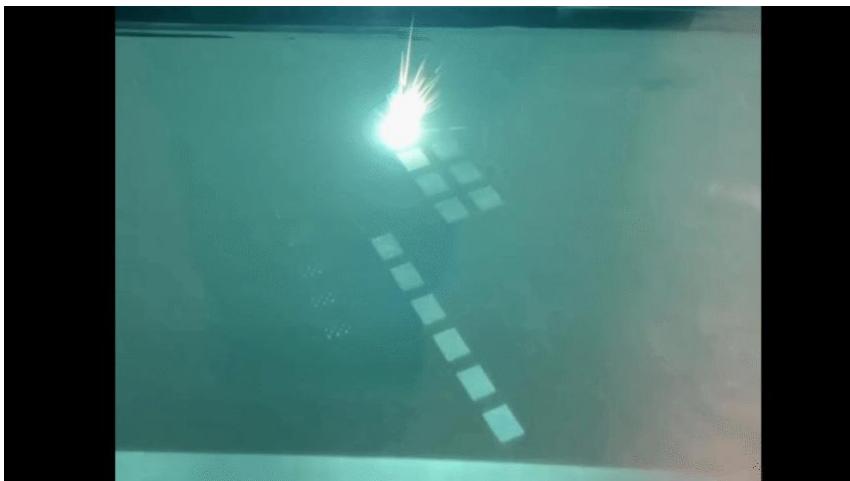
- Pristine cold-sprayed sample:
- ① Tantalum pentoxide at the surface
    - Natural passivating layer
  - ② D-implanted tantalum:
    - Peak shift:  
lower oxidation states?  
possible chemical interaction between Ta, D and O results in the formation of  $TaD_xO_y$
    - Peak shift for Ta → **Tantalum Deuteride?**
  - ③ Etching (2 keV Ar, 0.15 nm/sec) removes the oxide
    - D implantation depth estimate: < 24 nm?
  - ④ Thermal outgassing of D up to 1100 K
    - peaks positions restored
    - complete D outgas at a higher temperature

# Plasma-wetted surfaces in WHAM



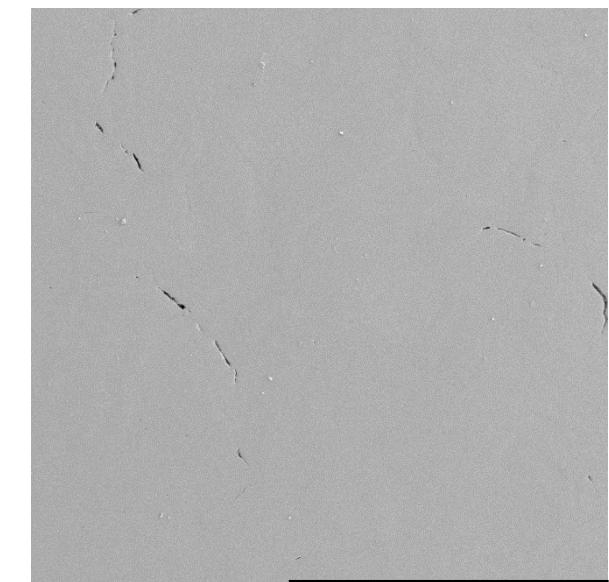
# 3D-printed tungsten

- UW  **ADDLab** is testing 3D-printed tungsten for fusion applications. First results are quite promising



100 µm

Particle interfaces (voids) or cracks



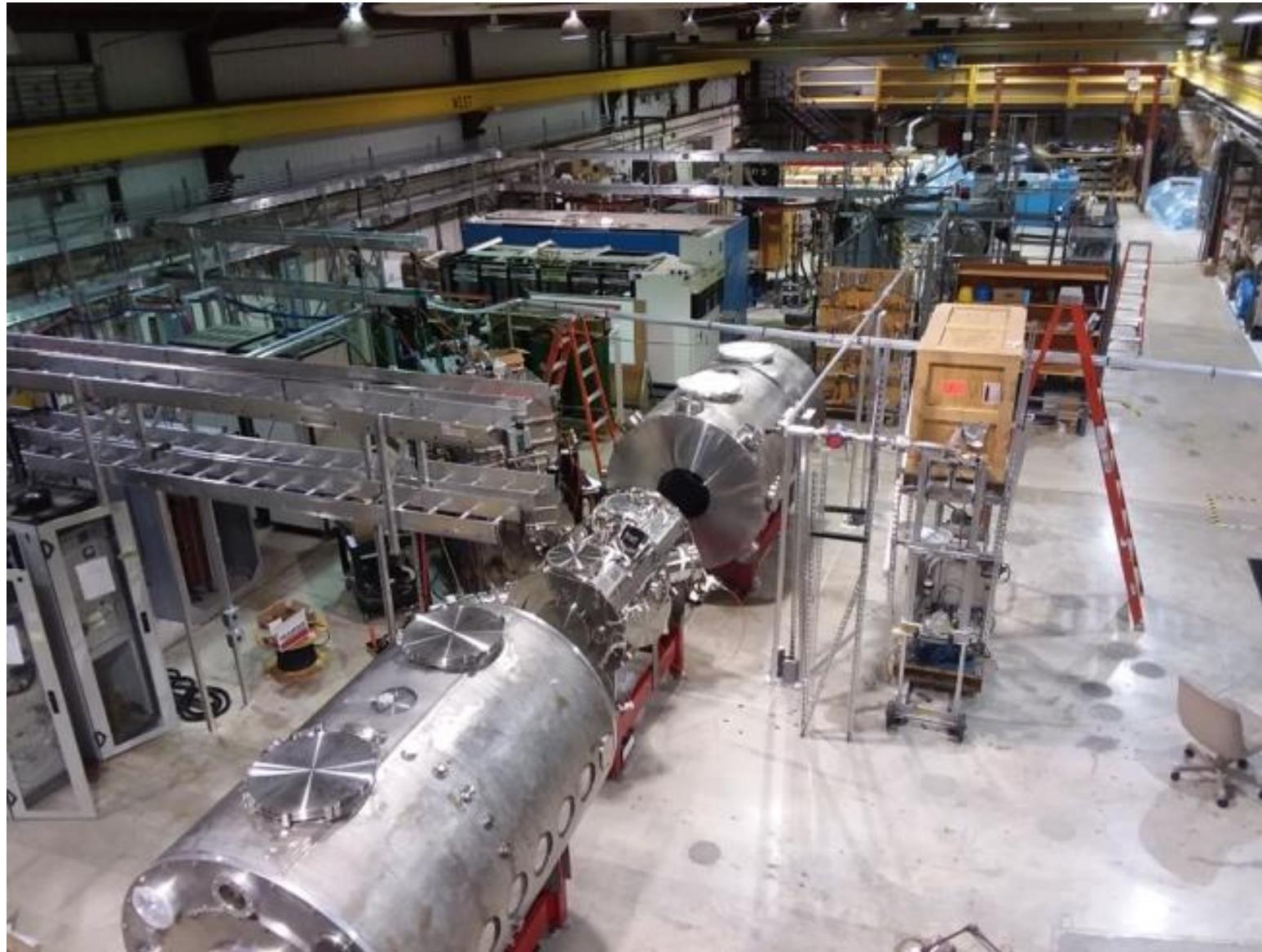
100 µm

Very few particle interfaces (and voids)

## **WHAM construction status update**

### **WHAM plasma-facing components**

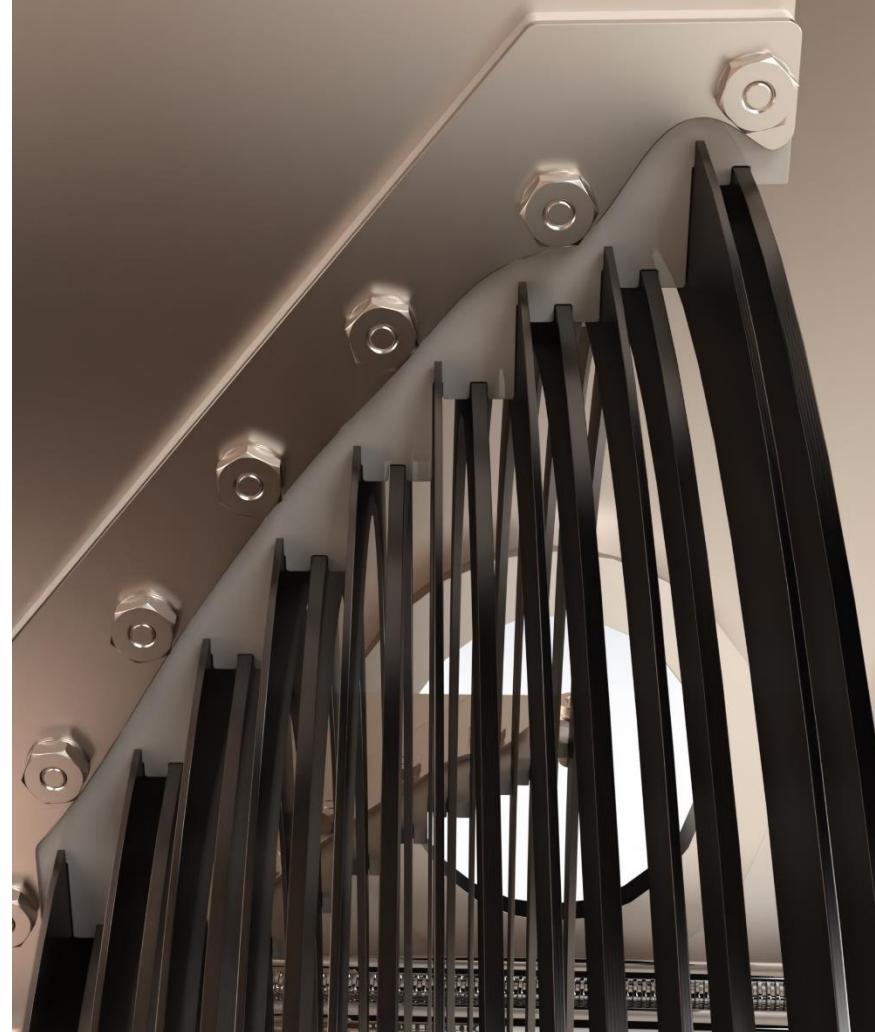
# WHAM construction update



# WHAM construction update: rings assembly

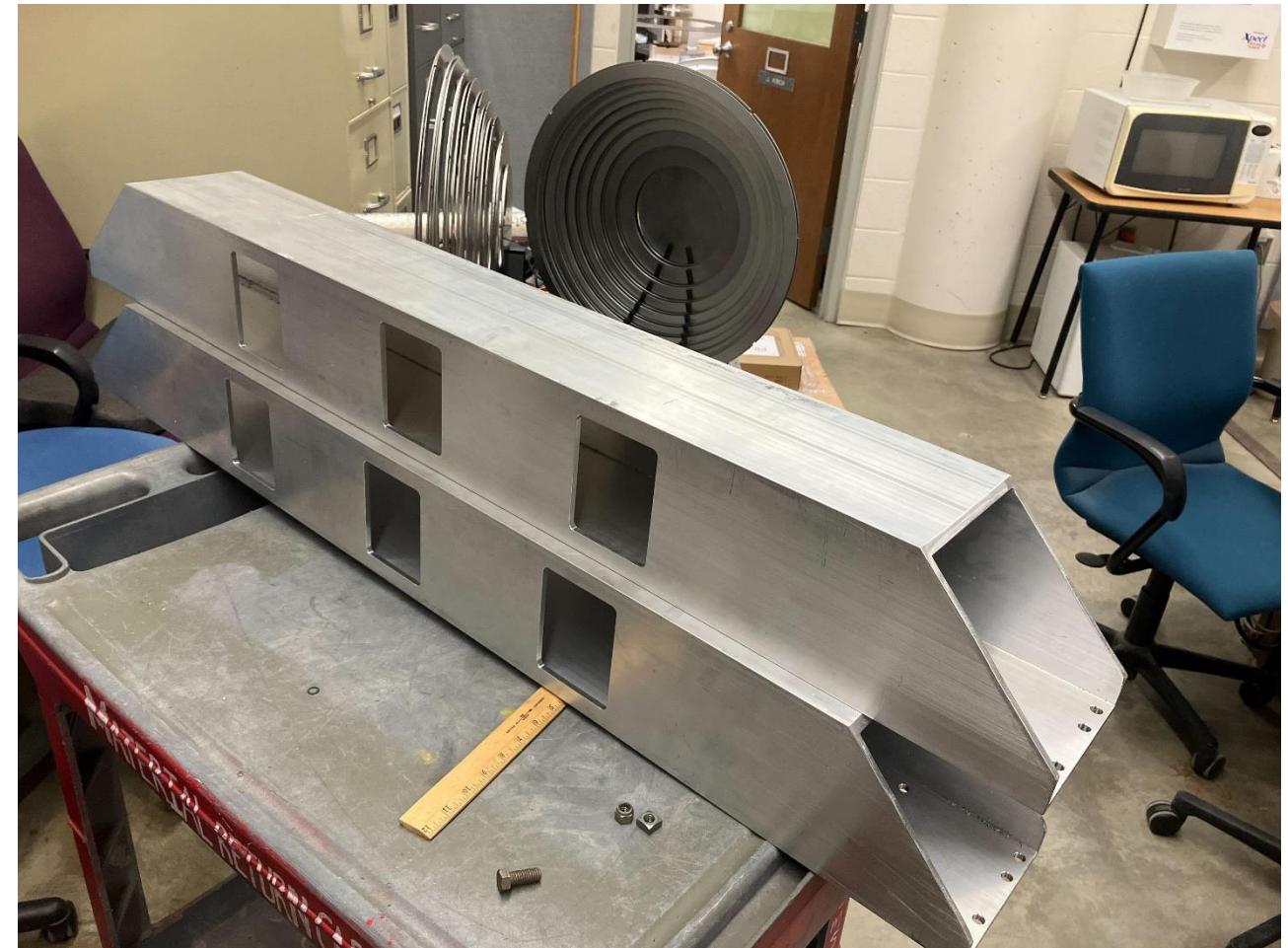


# WHAM construction update: rings assembly

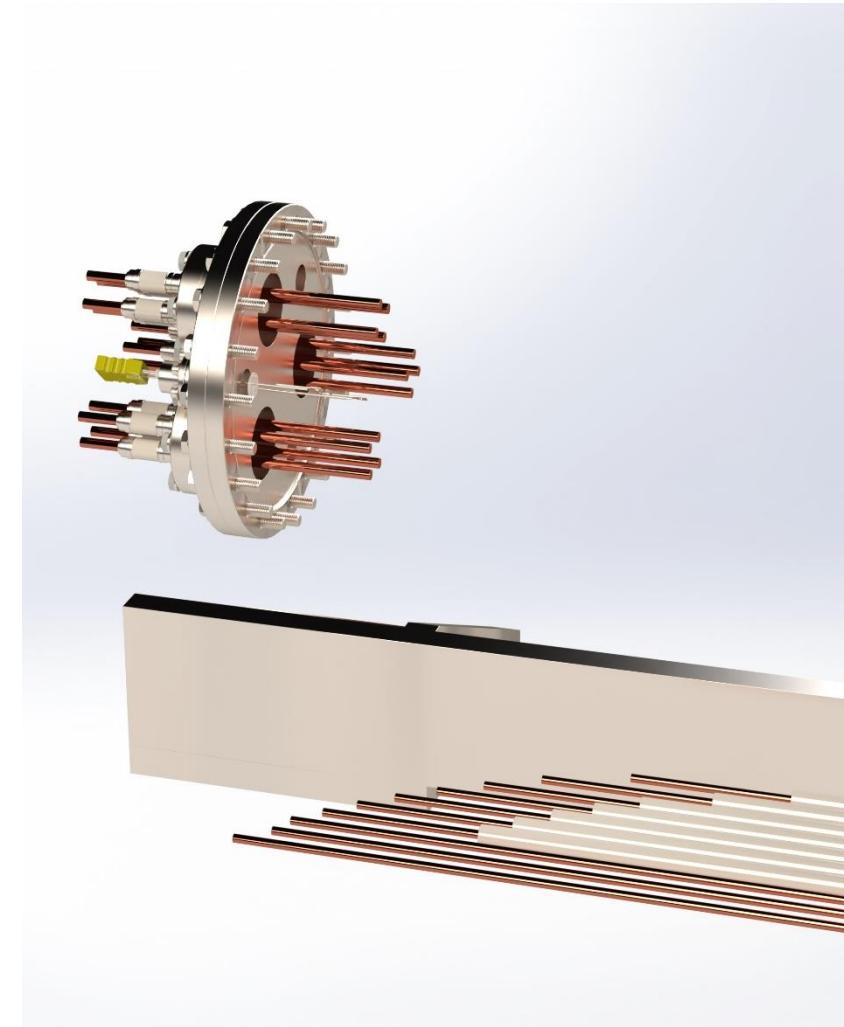
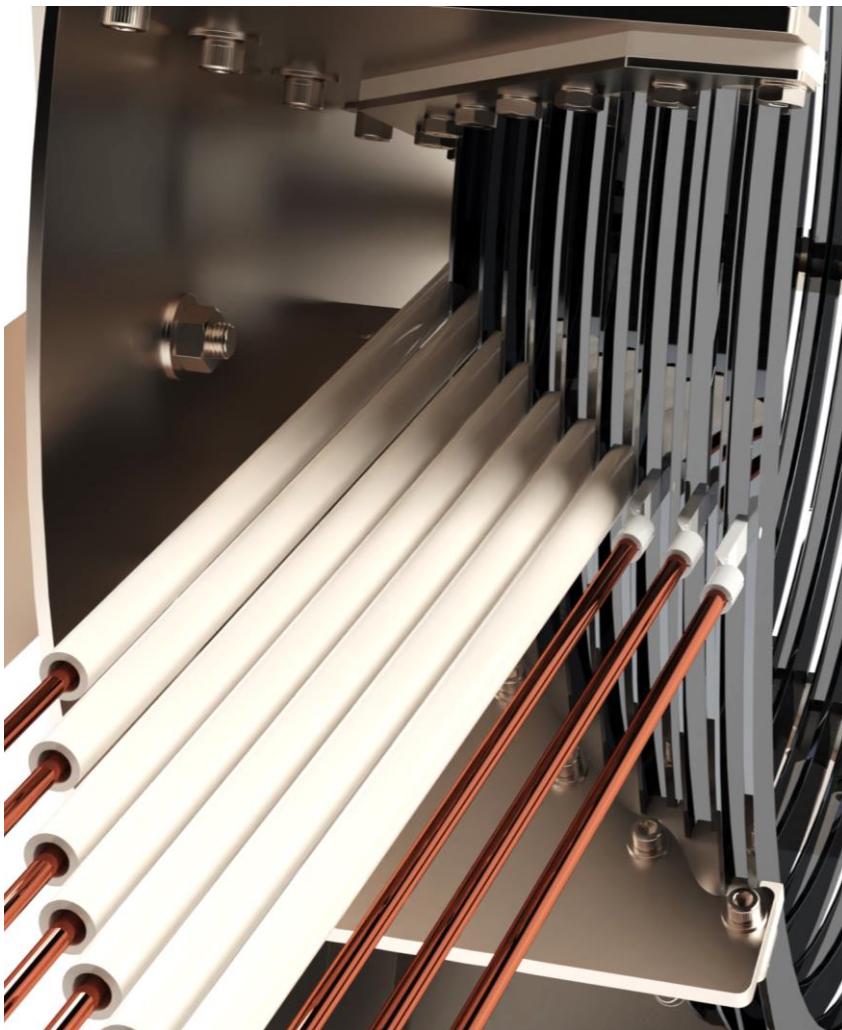


# WHAM construction update: rings assembly

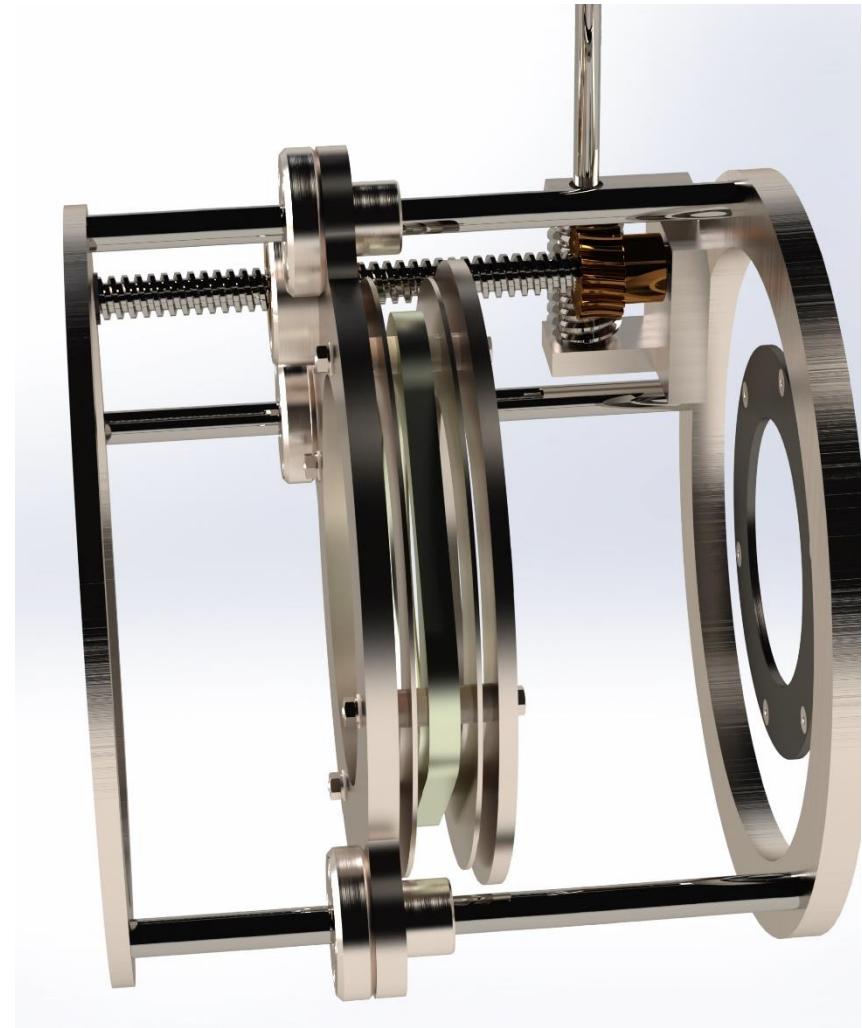
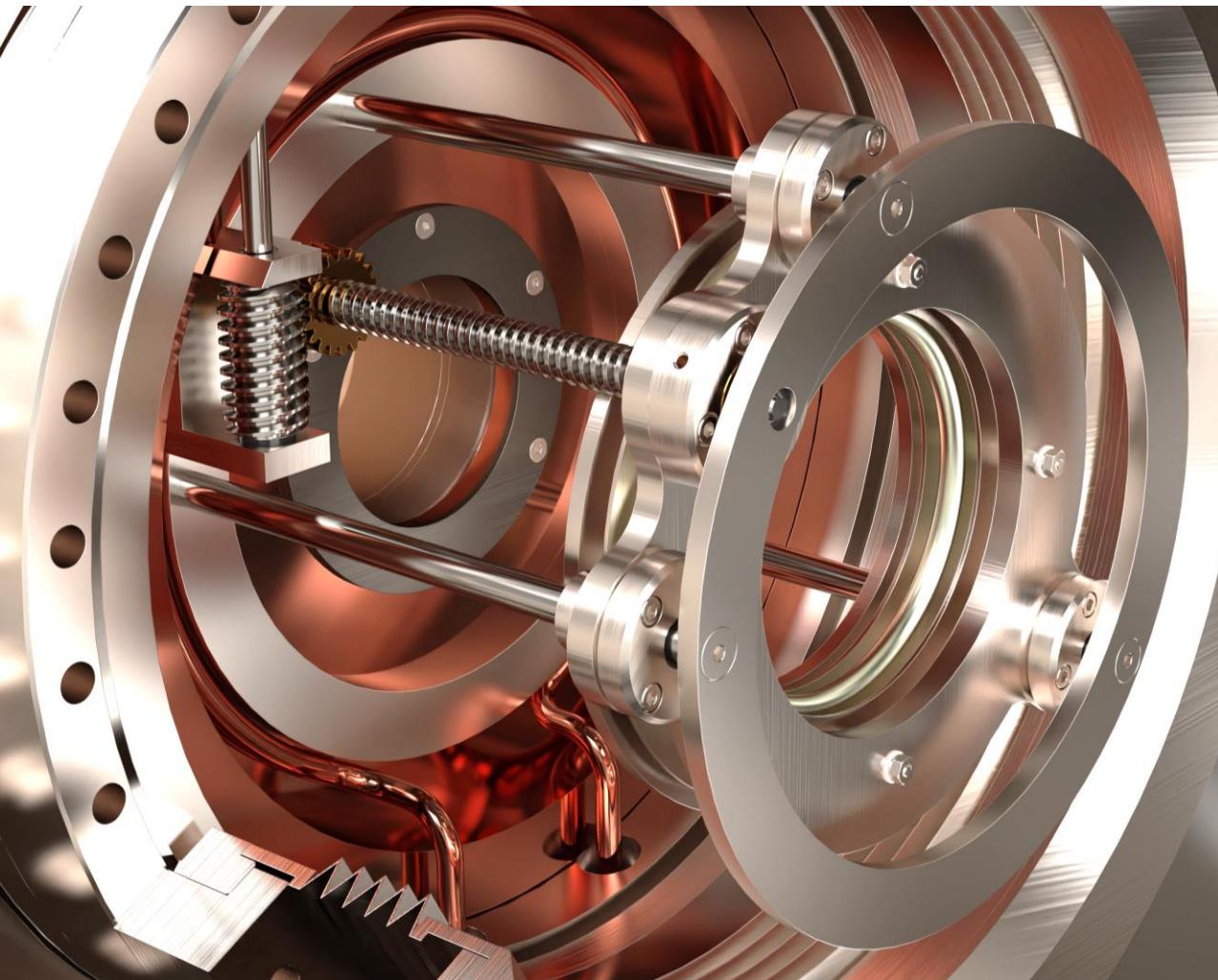
The ring assemblies will be mounted to these beams. The beams span across the diameter of the end cells and sit on a pair of linear rails.



# WHAM construction update: rings assembly



# WHAM construction update: limiter assembly



# WHAM construction update: limiter assembly



Do not forget to mention the 3D-printed W

# WHAM construction update: Argon glow discharge



All systems are being tested and the site is ready for arrival of the HTS magnets

# WHAM construction site timelapse



# Conclusions and Perspectives

- Small fusion devices require novel pumping strategies, ideally in-vessel absorbing walls
- Cold spray allows on site deposition and repair of PFCs
- High D trapping in tantalum: two orders of magnitude compared to sintered tungsten
- Enhanced D trapping in cold-spray Ta coating likely due to internal morphology
- Cold spray Ta coating can be regenerated (i.e., absorbed D releases by heating)



3dpsi.engr.wisc.edu  
ialovega@wisc.edu

## Perspectives and open questions:

- Development of additively manufactured tungsten and its alloys: tailoring PFM to the needs of FES
- Testing of cold spray components in real fusion experiments (WHAM and DIII-D), trial-error approach

P.S.: WHAM team is excited to have the first plasma this spring. Stay tuned!

# Extra slides